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1 **Nanoscale zero-valent iron (nZVI) immobilization onto graphene**
2 **oxide (GO)-incorporated electrospun polyvinylidene fluoride**
3 **(PVDF) nanofiber membrane for groundwater remediation via**
4 **gravity-driven membrane filtration**

5
6 **Jiawei Ren^{a,1}, Yun Chul Woo^{b,1}, Minwei Yao^a, Sungil Lim^a, Leonard D. Tijing^{a,*}, Ho**
7 **Kyong Shon^{a,*}**

8
9 ^aCentre for Technology in Water and Wastewater, School of Civil and Environmental
10 Engineering, University of Technology Sydney, P. O. Box 123, 15 Broadway, NSW 2007,
11 Australia

12 ^bDepartment of Land, Water and Environment Research, Korea Institute of Civil Engineering
13 and Building Technology (KICT), 283, Goyang-Daero, Ilsanseo-Gu, Goyang-Si, Gyeonggi-
14 Do, 10223, Republic of Korea

15
16 *Corresponding author: H. K. Shon (Hokyong.Shon-1@uts.edu.au)

17
18 ¹J. Ren and Y. C. Woo contributed equally to this work

24 **Abstract**

25 Nanoscale zero-valent iron (nZVI), with its high reactivity towards a broad range of
26 contaminants, has been a promising material for groundwater remediation. Membrane-
27 supported nZVI can both avoid nZVI agglomeration for better reactivity and recycle nZVI to
28 lower the risk of secondary pollution. In this study, we successfully fabricated a PVDF-GO
29 membrane via electrospinning technology and employed the functionalized nanofiber
30 membrane to immobilize nZVI particles. The addition of GO into PVDF nanofibers can both
31 increase the hydrophilicity to improve membrane flux and offer –COOH as a binder to
32 immobilize nZVI particles. PVDF-GO-nZVI membranes with different GO loadings (0%,
33 0.5%, 1%, 3% of PVDF) were tested with two typical nZVI-targeted contaminants (Cd(II) and
34 trichloroethylene (TCE)) via gravity-driven membrane filtration. The results show that
35 membrane with 1% GO had the best nZVI distribution against the aggregation and a better
36 performance in both Cd removal (100%) and TCE removal (82%). The nZVI membrane had
37 an high flux in gravity-driven filtration at 255LMH for Cd(II) and 265LMH for TCE
38 respectively. Generally, the developed PVDF-GO-nZVI electrospun nanofiber membrane had
39 an excellent performance in the gravity-driven membrane filtration system for groundwater
40 remediation.

41

42 **Keywords:** nano-scale zero valent iron, electrospun nanofiber membrane, graphene oxide,
43 groundwater remediation, gravity-driven membrane filtration

44

45

46 **1. Introduction**

47 With the rapid urban and economic development and the growing population,
48 contaminants in groundwater has become complex and intractable. Various pollutants like
49 heavy metals, chlorinated organics and dyes seriously influence the quality and safety of
50 groundwater (Vidmar et al., 2018). Nanoscale zero-valent iron particles (nZVI), with their high
51 reactivity towards a broad range of contaminants, are widely used in groundwater remediation
52 (Tosco et al., 2014). However, nZVI particles themselves are prone to agglomeration,
53 sedimentation and oxidization (Dong et al., 2017; Lefevre et al., 2016). Traditional methods to
54 address these issues are to add surfactants (Kim et al., 2010), suspending agents and stabilizers
55 (Hwang et al., 2014) to maintain the uniformity and activity of nZVI particles distributed in
56 the groundwater. Though nZVIs particles are effective in removing the target pollutants, the
57 adsorbed auxiliary additives and contaminants and the nZVI particles themselves may pose a
58 risk of secondary pollution as long as they remained in groundwater without retrieving them
59 (Dong et al., 2012).

60 To solve these issues, immobilizing the nZVI particles on a high-surface area membrane
61 substrate would be recycled and potentially reduce the risk of secondary pollution, thus this
62 approach of nZVI particle-loaded membrane is put forward here (Wan et al., 2017). An ideal
63 substrate for nZVI would be a membrane with high specific surface area, recyclable and easily-
64 functionalized properties, which not only can stably immobilize nZVI to avoid agglomeration
65 and sedimentation, but also can recycle nZVI particles or adsorbed contaminants after use (Ren
66 et al., 2017a). Electrospinning is one of the alternative membrane fabrication technologies but
67 has only found its application for nZVI immobilization in recent years due to its low cost, large
68 specific area, size-controllable and easily-functional properties (Ren et al., 2017b). PVA-PAA
69 membrane have been used to immobilize nZVI particles and the PVA-PAA-nZVI membrane

70 had an excellent removal on trichloroethylene (TCE) (Ma et al., 2012). Liu et al. (Liu et al.,
71 2014) also successfully immobilized nZVI particles onto a polyacrylonitrile (PAN)-based
72 oxidized membrane to reduce water contaminants, which had shown an excellent removal
73 performance. However, these kinds of membranes are lacking mechanical strength (Bichara et
74 al., 2014) or chemical stability (Manavi - Tehrani et al., 2010) and hard to be employed in a
75 long-term filtration process. Although, in our previous studies (Ren et al., 2017b; Ren et al.,
76 2019), we have successfully improved the immobilization of nZVI and mechanical strength of
77 the PVA-PAA-nZVI membrane, it still had some difficulties in nZVI regeneration due to lack
78 of chemical/ acid-base resistance.

79 Polyvinylidene fluoride (PVDF) with its strong mechanical strength and excellent
80 chemical stability has already been one of the most commonly-used materials in membrane
81 fabrication (Kang and Cao, 2014). However, PVDF is highly hydrophobic and does not have
82 active functional groups to bind with nZVI particles (Boo et al., 2016). To solve the problem,
83 graphene oxide (GO) with good hydrophilicity and a large number of carboxylic groups (–
84 COOH) could be a good additive to improve membrane hydrophilicity and introduce –COOH
85 onto PVDF membrane for nZVI immobilization (Wang et al., 2014a). Jang et al. (Jang et al.,
86 2015) fabricated PVDF-GO membrane by electrospinning and improved membrane flux due
87 to the introduction of GO in microfiltration process. However, the research on both using
88 PVDF-GO membrane for nZVI immobilization and its application via gravity-driven
89 membrane filtration process has not been reported yet.

90 Thus, this study aims to fabricate a PVDF-GO nanofiber membrane by electrospinning
91 and the fabricated membrane was used to immobilize nZVI particles by in-situ synthesis
92 method. The effect of GO on membrane fabrication and hydrophilicity were characterized by
93 scanning electron microscopy (SEM), attenuated total reflectance - fourier transform infrared
94 spectroscope (ATR-FTIR), X-ray diffraction (XRD), contact angle and mechanical strength.

95 The immobilization performance of nZVI was examined by SEM, energy dispersive X-ray
96 spectrometry (EDX), Thermo-gravimetric analysis (TGA), XRD and pore size. Two typical
97 nZVI-targeted contaminants (cadmium (Cd) and TCE, as representatives of heavy metal and
98 chloride organics) were separately filtered by the fabricated PVDF-GO-nZVI membranes via
99 gravity-driven filtration process. The effect of nZVI particles and contaminants on membrane
100 flux and removal performance were tested as well.

101 **2. Materials and methods**

102 **2.1 Materials**

103 PVDF (Kynar® 761, Mw = 441,000 g/mol) was purchased from Arkema InC., Australia.
104 Graphene oxide (GO), ferrous sulfate heptahydrate ($\text{FeSO}_4 \cdot 7\text{H}_2\text{O}$), sodium borohydride
105 (NaBH_4), cadmium nitrate ($\text{Cd}(\text{NO}_3)_2$), trichloroethylene (TCE), ethanol and n-hexane were
106 all purchased from Sigma-Aldrich. De-ionized (DI) water from a Millipore Milli-Q water
107 system was used; all the oxygen-free water was prepared by purging nitrogen gas for 10 min
108 to remove the dissolved oxygen and avoid the oxidation of nZVI particles. All chemicals were
109 used as received.

110 **2.2 Solution preparation and electrospinning of PVDF-GO nanofiber membranes**

111 GO particles with different ratios (**Table 1**) were added into a mixed solvent composed
112 of DMF and acetone (3:1 by wt%) at room temperature with sonication for 3 h to make it well-
113 distributed. Then PVDF powders were dissolved into the GO-mixed solvent by stirring for 48
114 h at room temperature.

115 The electrospinning set-up is similar to our previous study (Ren et al., 2017b). Nanofibers
116 were directly fabricated onto a rotating drum collector covered with baking paper in a sealed
117 chamber. The feed flow rate was controlled at 1 ml/h by a syringe pump. All the fabricated

118 mats were electrospun at an employed voltage of 18 kV and the tip-to-collector distance (TCD)
119 was 15 cm.

120 The polymer solution was supplied by a 12 ml syringe attached with a needle (21G, inner
121 diameter = 0.51 mm) connected via a metallic adaptor. The membranes thickness were
122 controlled at $100 \pm 5 \mu\text{m}$. The as-spun membranes were dried in oven at $60 \text{ }^\circ\text{C}$ for 48h to
123 remove the residual solvent.

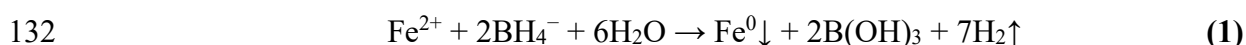
124 **Table 1.** Composition of PVDF-GO solution for electrospinning

Membrane	PVDF (wt%)	GO/PVDF (wt%)	DMF (wt%)	Acetone (wt%)
PVDF	15	0	68	17
PVDF-GO0.5		0.5		
PVDF-GO1		1		
PVDF-GO3		3		

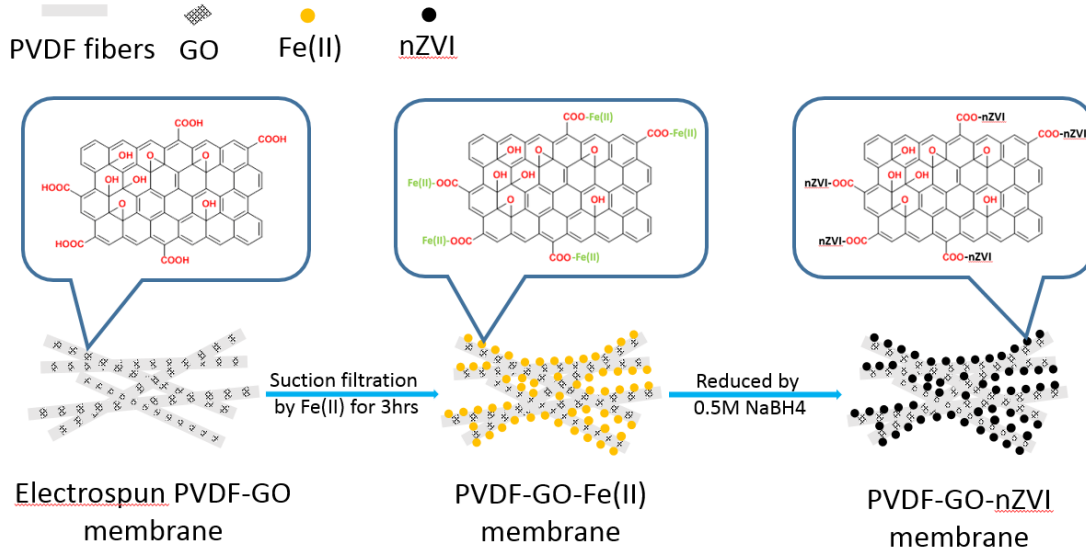
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126 2.3 Synthesis of nZVI on the PVDF-GO membrane

127 **Fig. 1** shows the schematic of the synthesis of nZVI-decorated PVDF-GO membranes.
128 All the PVDF-GO membranes were first filtered with 0.2 mol/L FeSO_4 solution for 3 h to allow
129 the complexation of ferrous ions (see **Fig. 1**). After that, all the PVDF-GO-Fe(II) membranes
130 were filtered with 0.5 mol./L NaBH_4 solution in an anaerobic chamber (filled by N_2 gas) to
131 form and immobilize nZVI particles. The reaction is shown below (Wang et al., 2008),



133 The successful formation of nZVI on the PVDF-GO membrane is indicated by the change
134 of color of the mat into black, and once the black color was stable (~10 mins generation time),
135 the PVDF-GO-nZVI mats were rinsed by ethanol then stored in a sealed glass bottle, which
136 was filled with nitrogen gas.



137

138 **Figure 1.** Schematic diagram of the nZVI particles growth and re-immobilization

139 **2.4 Gravity-driven membrane filtration**

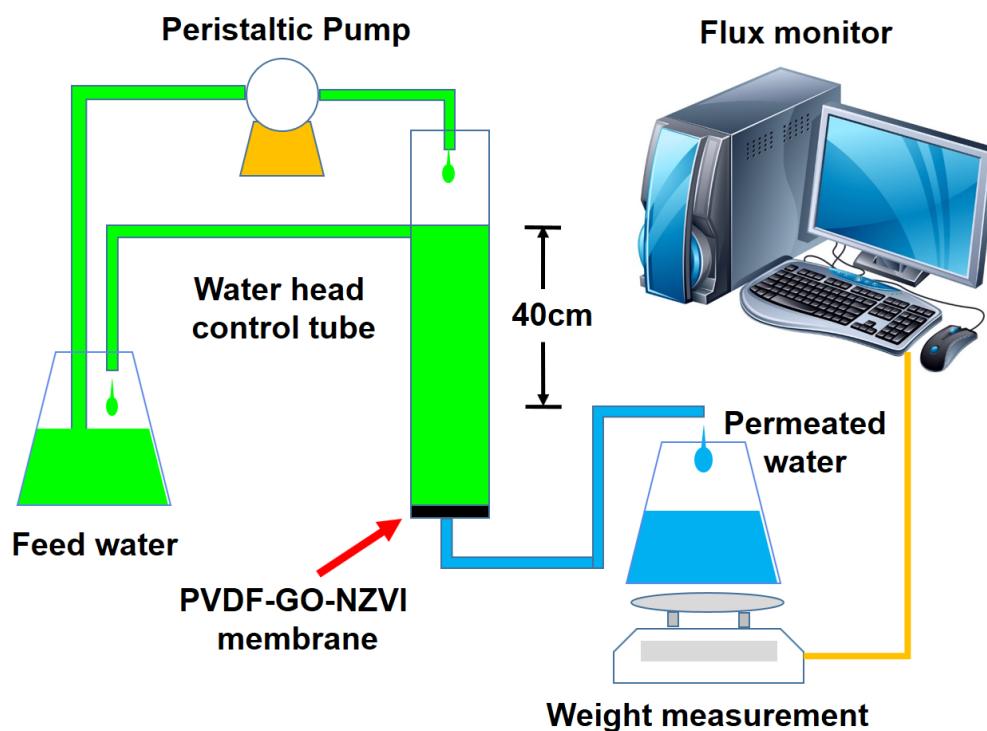
140 To test the performance of PVDF-GO-nZVI membranes on a relatively real condition of
 141 groundwater remediation, all the membranes were used to filtrate Cd(II) and TCE solutions
 142 separately by a gravity-driven membrane filtration application.

143 Schematic diagram of the filtration set-up is shown in **Fig. 2**. The thickness of each mat
 144 was controlled at $100 \pm 10 \mu\text{m}$ and the area for filtration was 14 cm^2 . The initial Cd(II)
 145 concentration was 20 mg/L and 100 ml solution was filtered in each filtration cycle (5 cycles
 146 for 500 mL in total). 10 mL permeated sample was withdrawn after each filtration cycle for the
 147 analysis of Cd(II) concentration via microwave plasma-atomic emission spectrometry (4100
 148 MP-AES, Agilent Technologies, US). Removal efficiency to Cd(II) was determined using the
 149 following equation:

150
$$\text{Removal efficiency (\%)} = \frac{C_{\text{int}} - C_{\text{end}}}{C_{\text{int}}} \times 100\% \quad (2)$$

151 where, C_{int} and C_{end} refer to the initial and final Cd(II) concentrations, respectively.

152



153

154 **Figure 2.** Schematic diagram of nZVI membrane gravity-driven filtration system

155 In addition, TCE as a typical chloride organic was also selected to test the dechlorination
 156 capacity of PVDF-GO-nZVI membrane. Similarly, the concentration of TCE was also 20 mg/L
 157 but the permeated sample was withdrawn and transferred to a new vial for the subsequent
 158 liquid-liquid extraction. The chemical n-hexane was used to extract TCE from water. The
 159 concentration of the extracted TCE solution was measured by GC-MS-TQ8040 (Shimadzu,
 160 Japan). The removal efficiency was calculated using the following equation:

161
$$\text{TCE removal efficiency (\%)} = \frac{C_{\text{int}} - C_{\text{end}}}{C_{\text{int}}} \times 100\% \quad (3)$$

162 where, C_{int} and C_{end} are the initial and the final TCE concentrations, respectively.

163 **2.5 Membrane Characterizations**

164 Membrane surface morphology and element component analysis were measured by scanning
 165 electron microscopy and energy dispersive X-ray spectrometry (SEM and EDX, Zeiss Supra
 166 55VP, Carl Zeiss AG). All the fiber/particle sizes and distributions were measured by Image J.

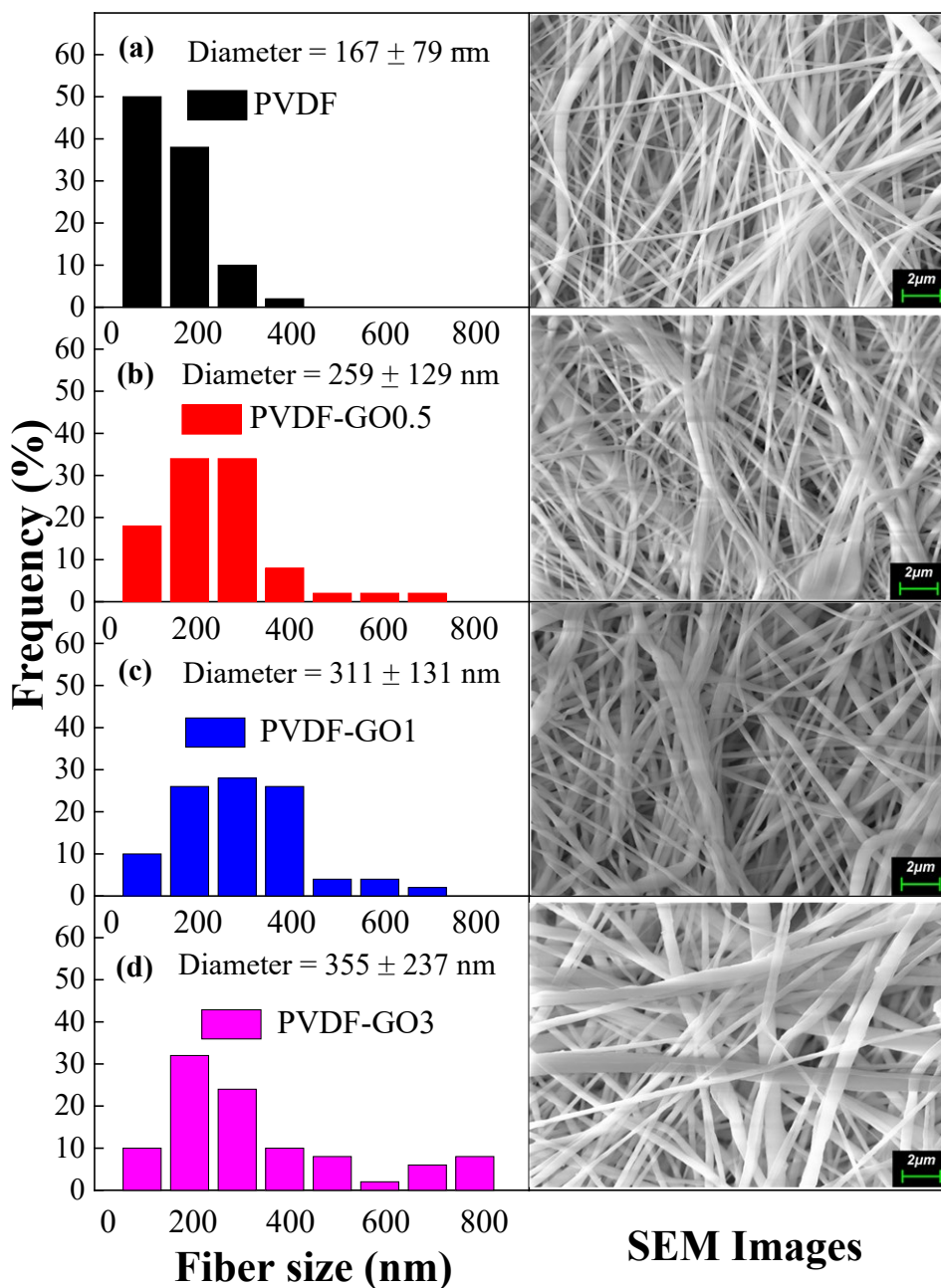
167 Thermo-gravimetric analysis (TGA) was measured by Discovery TGA thermo-gravimetric
168 analyzer (SDT-Q600, US) from 80°C to 800°C at a heating rate of 10°C/min in N₂ atmosphere
169 [27]. The functional groups of membranes/particles were analyzed by attenuated total
170 reflectance Fourier transform infrared spectroscopy (ATR-FTIR) via Paragon 1000
171 Spectrometer (PerkinElmer, USA), wavelength from 600 to 4000 cm⁻¹ with a minimum of 40
172 scans. X-Ray diffraction (XRD) (Siemens D5000) was carried out over Bragg angles ranging
173 from 1° to 80° (Cu K α , $\lambda=1.54059\text{\AA}$) (Woo et al., 2016). The mechanical properties of different
174 membrane samples were measured using a Universal Testing Machine (UTM, Lloyd),
175 equipped with a 1 kN load cell. The test was conducted using a constant elongation velocity of
176 5 mm/min under room temperature (Woo et al., 2017). Galwick (surface tension of 15.9 mN/m)
177 was used to wet the membranes in pore size. Each sample was first applied with N₂ gas to
178 measure the gas permeability. A sessile drop method utilizing Theta Lite 100 (Attension,
179 Sweden) with built-in software was used to analyses the contact angles of all the membranes
180 (Yao et al., 2019). A water droplet around 5 μl was released from a needle tip onto the
181 membrane surface. A motion camera was mounted to take photos at a rate of 12 frames per
182 second. Each sample was measured for three times and the average value was taken.

183 **3. Results and discussion**

184 **3.1 Effect of GO on PVDF membranes**

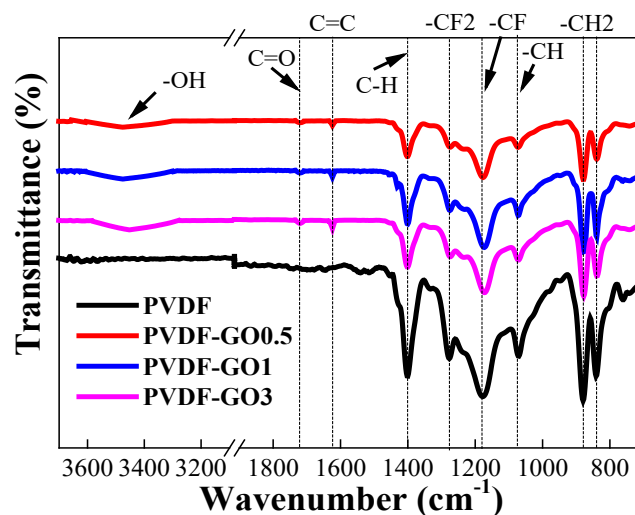
185 **Fig. 3** shows the SEM images and fiber size distributions of as-spun PVDF-GO nanofiber
186 membranes with different GO ratios. All the membranes showed bead-less and defect-free
187 cylindrical nanofiber structure. The average fiber sizes of the as-spun membranes (PVDF,
188 PVDF-GO0.5, PVDF-GO1 and PVDF-GO3) were 167 ± 79 , 259 ± 129 , 311 ± 131 and $355 \pm$
189 237 nm, respectively. It is clear that the fiber size increased with the increase in the GO added

190 into the PVDF solution. However, the addition of GO broke the continuity of polymer jetting
 191 during the electrospinning, which resulted to a wider fiber size distribution. Similar results
 192 were also reported in previous studies (Jang et al., 2015).



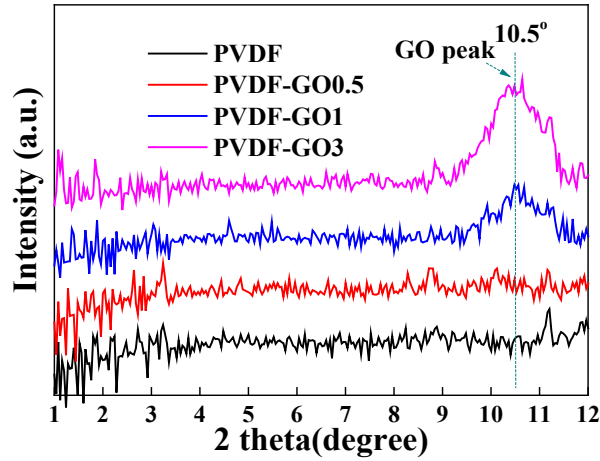
193
 194 **Figure 3.** Surface SEM images and corresponding fiber size distributions of (a) PVDF; (b)
 195 PVDF-GO0.5; (c) PVDF-GO1 and (d) PVDF-GO3 electrospun nanofiber membranes

196 **Fig. 4** shows the ATR-FTIR spectra of the PVDF-GO membranes. For the pristine PVDF,
 197 the characteristic absorption band is mainly in the wavenumber region of 840 – 1402 cm^{-1} .
 198 Specifically, the peak intensities at 1181, 1275 cm^{-1} represents symmetrical stretching of $-\text{CF}$
 199 and $-\text{CF}_2$ stretching, while peaks at 840, 883, 1070 and 1402 cm^{-1} represents the $-\text{CH}_2$ rocking,
 200 $-\text{CH}_2$ wagging of vinylidene, bending of $-\text{CH}$ bond and the stretching vibration of C-H bonds
 201 (Ghaffar et al., 2018; Zeng et al., 2016) respectively. In terms of GO membranes, The peaks at
 202 1622, 1720 and 3500 cm^{-1} indicated the existence of C=C, $-\text{COOH}$ and $-\text{OH}$, which proved the
 203 deposition of GO onto the PVDF membrane (Ghaffar et al., 2018). With the increase of GO
 204 ratio, the peak of $-\text{COOH}$ and C-OH increased correspondingly but the characteristic peaks of
 205 PVDF reduced due to the ratio of PVDF in the composite reduced as well. The XRD results of
 206 all the electrospun membranes are presented in **Fig. 5**. There are clear peaks at $2\theta = 10.5^\circ$ on
 207 all the PVDF-GO membranes indicating the existence of GO. The peak intensity increased
 208 with the increase of GO/PVDF ratio and PVDF-GO3 had the highest GO peak intensity, which
 209 was consistent with FTIR results (**Fig. 4**).



210

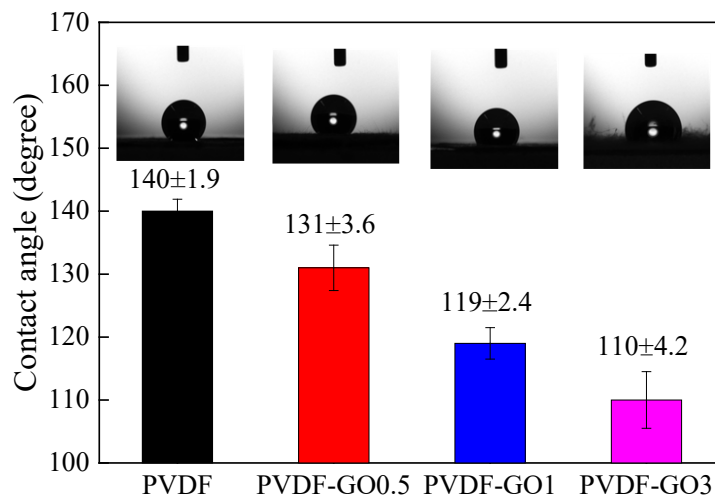
211 **Figure 4.** ATR-FTIR spectra of all the PVDF-GO electrospun nanofiber membranes



212

213 **Figure 5.** XRD patterns of all the PVDF-GO electrospun nanofiber membranes

214 A contact angle measurement was conducted on the PVDF-GO membranes and the results
 215 are shown in **Fig. 6**. The pristine PVDF membrane showed the highest contact angle of $140 \pm$
 216 1.9° , indicating being highly hydrophobic. Expectedly, the addition of GO nanoparticles
 217 significantly reduced the contact angle of PVDF membrane, because the $-\text{COOH}$ and $-\text{OH}$ on
 218 GO are hydrophilic. With the increase of GO ratio, the average contact angle reduced from
 219 140° to 110° and the PVDF-GO3 had the lowest contact angle at $110 \pm 4.2^\circ$.



220

221 **Figure 6.** Contact angles of all the PVDF-GO electrospun nanofiber membranes

222

223 **Table 2.** Mechanical properties of the neat PVDF and PVDF-GO electrospun nanofiber
224 membranes

	Tensile strength (MPa)	Strain (%)	Young's modulus (MPa)
PVDF	6.4	49.3	13
PVDF-GO0.5	5.2	27	19.3
PVDF-GO1	4.8	12.6	38.1
PVDF-GO3	4.4	9.5	46.3

225
226 The mechanical properties of the neat PVDF membrane and the PVDF-GO membranes are
227 presented in **Table 2**. It is obvious that the addition of GO led to a decrease of the tensile
228 strength as well as the percentage of the strain. With the increase of GO ratio in the
229 membrane, the tensile strength decreased from 6.4 MPa to 4.4 MPa and the strain reduced
230 from 49.3% to 9.5%. However, the Young's modulus increased from 13 MPa to 46.3 MPa,
231 which means the more GO added, the more plastic the nanocomposites will be. Although the
232 PVDF-GO membranes became more plastic, they had a better resistance to deformation of
233 fiber structure (Ren et al., 2019).

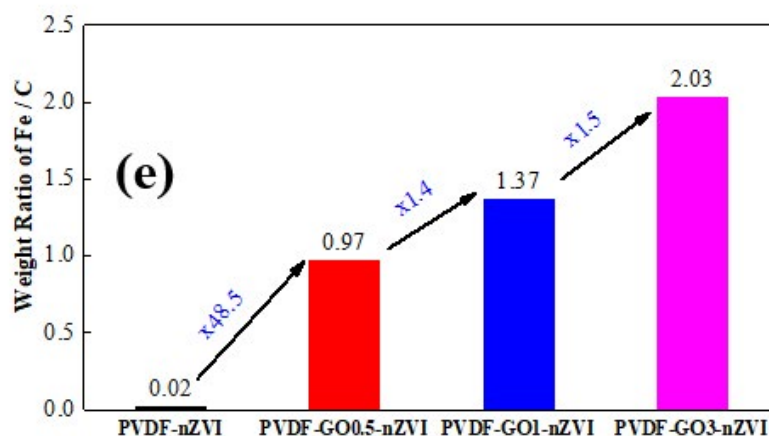
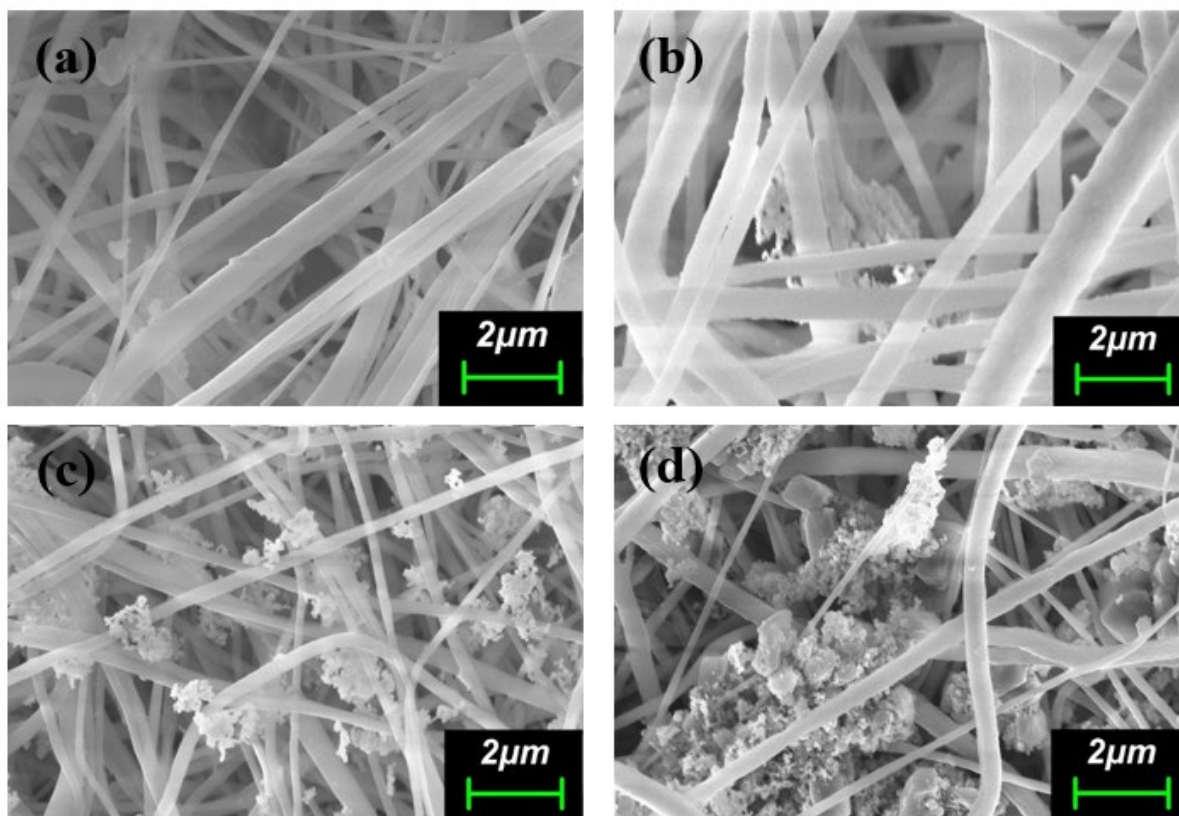
234 3.2 Immobilization of nZVI onto PVDF-GO membrane

235 The surface SEM images of all the nZVI-immobilized PVDF-GO membranes are shown
236 in **Fig.7**. It is clear that nZVI nanoparticles have been successfully generated on each PVDF-
237 GO membrane. However, the neat PVDF membrane without GO did not show a good
238 immobilization of nZVI due to the lack of functional groups and only a few nZVI particles
239 were stuck among fibers via physical interaction other than chemical bonding. With the
240 increase of GO content in PVDF nanofiber, nZVI immobilized on PVDF-GO membranes
241 increased as well. It is noticed that although PVDF-GO3 seemed to have the most nZVI
242 particles immobilized onto nanofibers, the particles were more aggregated and not well-
243 distributed (**Fig. 7c**). Specifically, nZVI particles were more immobilized on the thin fibers
244 than thicker fibers. This is due to the fact that the functional moieties (-COOH) are mainly on

245 the edge of GO and the edges of GO are more likely to pierce the surface of thinner fibers
246 rather than encapsulated inside the thicker fiber (Hu et al., 2016). **Fig. 7e** shows the Fe/C weight
247 ratio of each nZVI membrane by EDS analysis in the SEM images and the results are highly
248 consistent with the SEM images. The Fe/C ratio of neat PVDF was extremely low at 0.02,
249 approximately 1/50 lower than those on PVDF-GO membranes, which indicates that the neat
250 PVDF membrane failed to immobilize nZVI particles. In contrast to PVDF-GO membranes,
251 with the ratio of GO added into PVDF membrane, the Fe/C ratio increased significantly.
252 PVDF-GO3-nZVI (**Fig. 7d**) had the highest Fe/C ratio at 2.03 on its membrane surface, around
253 1.5 times higher than that on PVDF-GO1-nZVI (**Fig. 7c**).

254

255



256

257 **Figure 7.** Surface SEM images of: (a) PVDF-nZVI; (b) PVDF-GO0.5-nZVI; (c) PVDF-GO1-
 258 nZVI; (d) PVDF-GO3-nZVI electrospun nanofiber membranes; and (e) the weight ratio of
 259 Fe/C on the surface of the four nZVI-immobilized electrospun nanofiber membranes based
 260 from EDS analysis.

261 The pore size of PVDF-GO membranes before and after nZVI immobilization are
 262 presented in **Table 3**. Before the immobilization of nZVI, the mean pore size increased with
 263 the GO content (from 0.8 μm to 1.21 μm) due to the increase of fiber size (Wang et al., 2014b).

264 After nZVI immobilization, all the membrane pore sizes decreased and the pore size of the
 265 membrane with a higher GO ratio reduced more significantly. This is mainly because the nZVI
 266 particles were deposited on the surface of fibers and then occupied the voids among the fibers.
 267 Membrane with high GO ratio immobilized more nZVI particles and hence the pore size
 268 reduced more.

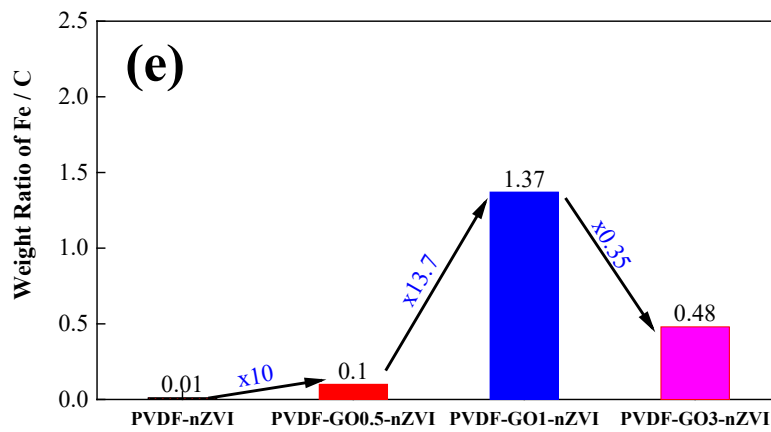
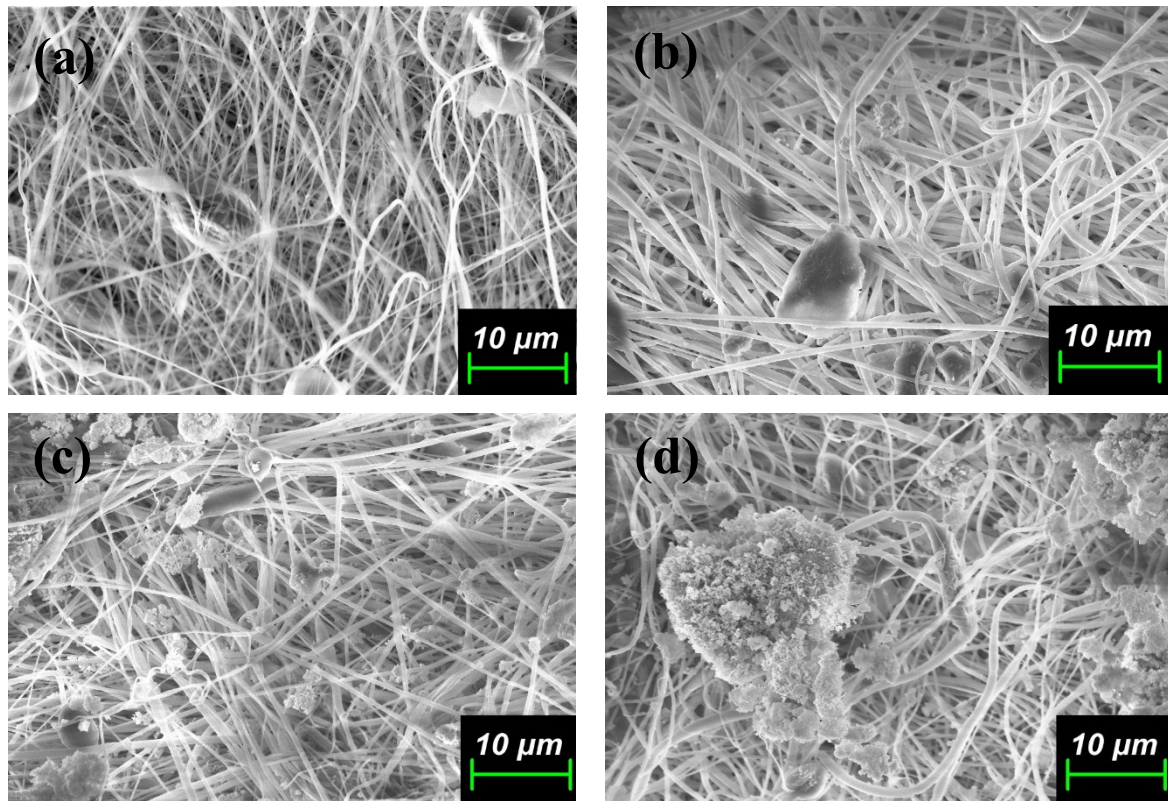
269 **Table 3.** Mean pore sizes of PVDF-GO membranes before/after nZVI immobilization

	Mean pore size (μm)		Mean pore size (μm)
PVDF	0.8	PVDF-nZVI	0.79
PVDF-GO0.5	0.99	PVDF-GO0.5-nZVI	0.95
PVDF-GO1	1.13	PVDF-GO1-nZVI	1.06
PVDF-GO3	1.21	PVDF-GO3-nZVI	1.11

270

271 The cross-sectional SEM images of all the nZVI-immobilized PVDF-GO membranes and their
 272 Fe/C ratios are shown in **Fig. 8**. Only very few nZVI particles immobilized onto the neat PVDF-
 273 nZVI (**Fig. 8a**) and PVDF-GO0.5-nZVI (**Fig. 8b**) membranes. Correspondingly, their Fe/C
 274 ratios were lower than those on their surfaces. This is mainly caused by the stronger
 275 hydrophobicity (higher contact angles, **Fig. 6**) and the smaller mean pore sizes of PVDF and
 276 PVDF-GO1 (0.8 and 0.99 μm respectively, **Table 3**). After forming nZVI particles on the
 277 membrane surface, the mean pore size became smaller and reducing agent cannot evenly
 278 penetrate the hydrophobic PVDF and PVDF-GO0.5 membranes. As a result, only few nZVI
 279 particles were immobilized inside the membrane. However, the immobilization of nZVI
 280 particles inside PVDF-GO1-nZVI (**Fig. 8c**) was as much as the nZVI on its surface (**Fig. 7c**),
 281 which means the nZVI particles were uniformly distributed on the whole PVDF-GO1-nZVI.
 282 Meanwhile, its Fe/C weight ratio on cross-section (**Fig. 8e**) was also same as the result on
 283 surface measurement (**Fig. 7e**), which proved the high uniformity of nZVI immobilized onto
 284 the PVDF-GO1-nZVI as well. In terms of PVDF-GO3-nZVI, due to the aggregation of nZVI

285 on the surface (**Fig. 7d**), its pore size decreased significantly by nZVI blocking and hence
 286 hindering the generation of nZVI inside the membrane, leading to a lower Fe/C ratio inside.

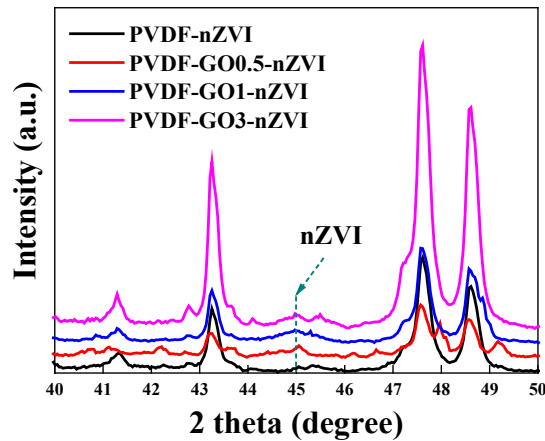


287

288 **Figure 8.** Cross-sectional SEM images of (a) PVDF-nZVI; (b) PVDF-GO0.5-nZVI; (c) PVDF-
 289 GO1-nZVI; (d) PVDF-GO3-nZVI electrospun nanofiber membranes and (e) the weight ratio
 290 of Fe/C on the cross-sectional of the four nZVI immobilized electrospun nanofiber membranes

291 XRD results of nZVI-incorporated PVDF-GO electrospun nanofiber membranes are
 292 presented in **Fig. 9**. The peak at 45° represents the crystal structure of α -Fe (110), indicating

293 the existence of nZVI onto PVDF-GO membranes (Gu et al., 2018), although the intensity of
294 the peak was not very high due to a less mass ratio of nZVI on the nanocomposites and the
295 strong peak of amorphous structure of PVDF-GO electrospun nanofiber membranes.



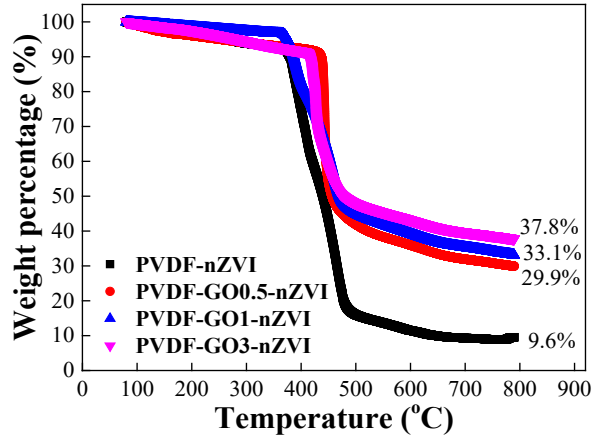
296

297 **Figure 9.** XRD pattern of all the PVDF-GO-nZVI electrospun nanofiber membranes

298 TGA was used to characterize the loading capacity of nZVI particles immobilized onto the
299 dual-crosslinked nanofiber membranes (**Fig. 10**). The weight loss on each membrane on the
300 temperature range from 150 to 400°C indicates the decomposition of GO (Issa et al., 2016;
301 Şinoforoğlu et al., 2013). Afterwards, the major weight loss at the region from 400 to 500°C is
302 attributed to the decomposition of the PVDF polymers(Thakur et al., 2011). Obviously, the
303 neat PVDF without GO decreased the most weight percentage.

304 At 800°C, all the polymeric nanofibers had already been decomposed, leaving carbons and
305 nZVI particles as the residual mass (Sharma et al., 2015). Neat PVDF remained only carbon
306 and few nZVI particles at 9.6 wt%. PVDF-GO3 immobilized the most nZVI with about 37.8
307 wt%, while PVDF-GO1 immobilized the second most nZVI around 33.1 wt%, around 28.2%
308 and 23.5% higher than the neat PVDF. Basically, PVDF with a higher GO ratio immobilized
309 more nZVI particle, which is consistent with SEM results.

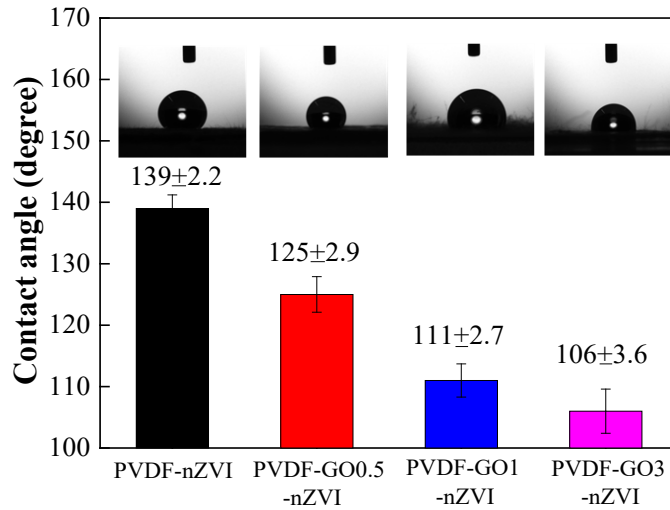
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Figure 10. TGA of all the PVDF-GO-nZVI electrospun nanofiber membranes



313

314

Figure 11. Contact angles of all the PVDF-GO-nZVI electrospun nanofiber membranes

315

Contact angles after nZVI immobilization on the membranes are presented in **Fig. 11**.

316

All the membranes reduced contact angle after nZVI immobilization due to the hydrophilicity

317

of the oxide shell on the nZVI surface. With the least nZVI immobilized, the PVDF-nZVI

318

remained the highest contact angle at $139 \pm 2.2^\circ$. There is relative larger improvement of

319

hydrophilicity on PVDF-GO membranes. The contact angle of PVDF-GO0.5-nZVI was $125 \pm$

320

2.9° , around 6° lower than that before nZVI immobilization. PVDF-GO1-nZVI was improved

321

the most, whose contact angle reduced by 8° from $119 \pm 2.4^\circ$ to $111 \pm 2.7^\circ$. However, the

322

contact angle of PVDF-GO3-nZVI only decreased around 4° which did not show a better

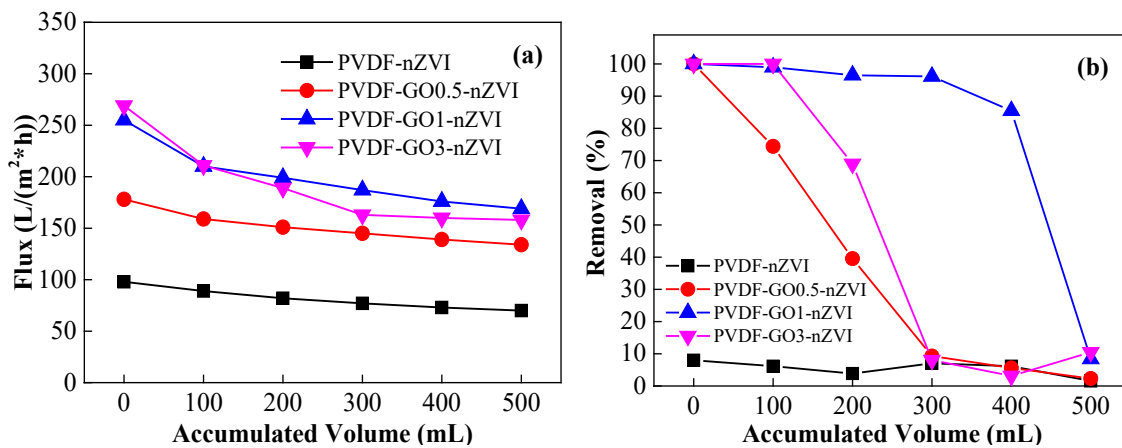
323 improvement than PVDF-GO1-nZVI. Although PVDF-GO3 membrane immobilized the most
324 amount of nZVI particles, those nZVI particles are more aggregated, leading to increase surface
325 roughness, which can only improve the hydrophilicity on the part where the nZVI particles
326 gathered.

327 **3.3 Performance of PVDF-GO-nZVI membrane in removing contaminants via** 328 **gravity-driven membrane filtration**

329 All the nZVI-immobilized electrospun nanofiber membranes were tested via gravity-
330 driven membrane filtration, with water head controlled at 0.4 m. Two typical nZVI-targeted
331 contaminants (Cd(II) and TCE) were employed to test the permeability and removal
332 performance of nZVI-immobilized electrospun nanofiber membranes.

333 **Fig. 12** shows the evolution of flux and Cd(II) removal with the increase of filtration
334 volume. The flux of each membrane decreased with the filtration of Cd(II) solution (**Fig. 12a**).
335 PVDF-nZVI (i.e., without GO) had the lowest flux (from 98 LMH to 70 LMH) among the four
336 membranes due to the smallest membrane pore size and the strongest hydrophobicity. In
337 contrast, with the increase of GO content, the mean pore size and the hydrophilicity of PVDF-
338 GO membranes increased remarkably, thereby contributing to a better permeability. PVDF-
339 GO3-nZVI had the highest initial flux at 269 LMH followed by PVDF-GO1-nZVI at 255 LHM.
340 However, the flux of PVDF-GO3-nZVI declined significantly. After 100 mL filtration, the flux
341 of PVDF-GO3-nZVI was lower than that of PVDF-GO1-nZVI. This is because, with the
342 adsorption of Cd(II) on the surface of nZVI particles, the size-increased nZVI made the
343 membrane pore volume and pore size reduced. Furthermore, nZVI particles on the PVDF-GO3
344 are more aggregated and hence the pore size reduced more significantly. In **Fig. 12b**, the
345 PVDF-nZVI had a very little removal lower than 8% over the whole filtration test while all the
346 PVDF-GO-nZVI membranes had 100% initial removals of Cd(II). PVDF-GO1-nZVI kept the

347 excellent removals higher than 90% within 300ml filtration test and after 400 ml, it started to
 348 decline and eventually became lower than 10%. However, PVDF-GO0.5-nZVI and PVDF-
 349 GO3-nZVI started to decline since 100 ml and 200 ml respectively. After 300 ml filtration,
 350 both their removals were lower than 10%. Compared with PVDF-GO1-nZVI, on one hand,
 351 PVDF-GO0.5-nZVI did not immobilize enough nZVI to adsorb more Cd(II); On the other hand,
 352 although PVDF-GO3-nZVI successfully immobilized more weight of nZVI on the surface,
 353 most of them were too aggregate to expose to Cd(II) and hence failed to remain a high removal.
 354 The main mechanism of Cd(II) removal by nZVI is chemisorption (bidentate inner-sphere
 355 surface complexation) (Boparai et al., 2013). Therefore, Cd(II) was adsorbed on the surface of
 356 nZVI in the form of ions, specifically CdFe₂O₄ reported in previous study (Hidalgo et al., 2015).
 357 With the increase of filtration volume, the reaction sites of nZVI was covered by CdFe₂O₄ (Fig.
 358 14a) and hence the removal reduced. As a result, the particle size should increase and lead to a
 359 decline of flux.

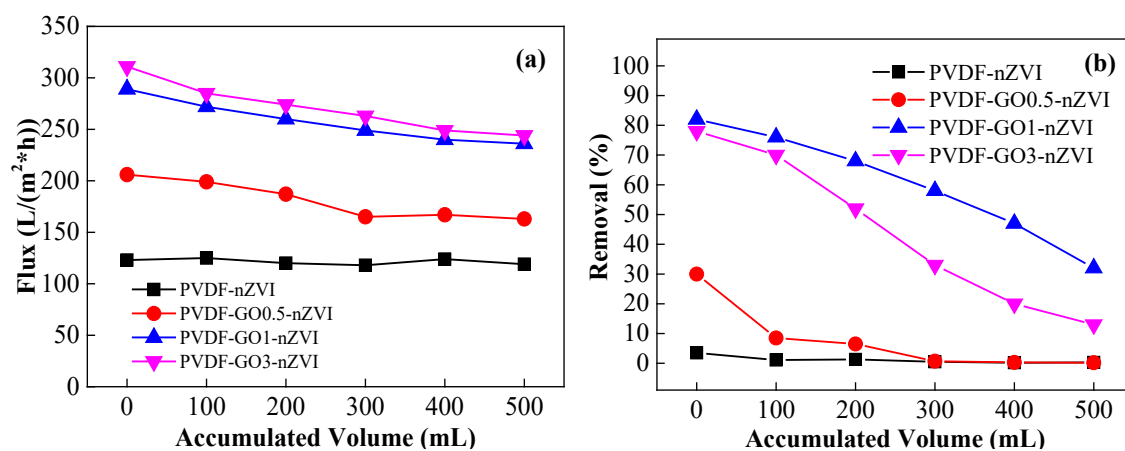


360
 361 **Figure 12.** Performances of PVDF-GO-nZVI electrospun nanofiber membranes in gravity-
 362 driven membrane filtration test for Cd(II) removal: (a) Flux and (b) Cd(II) removal ratio

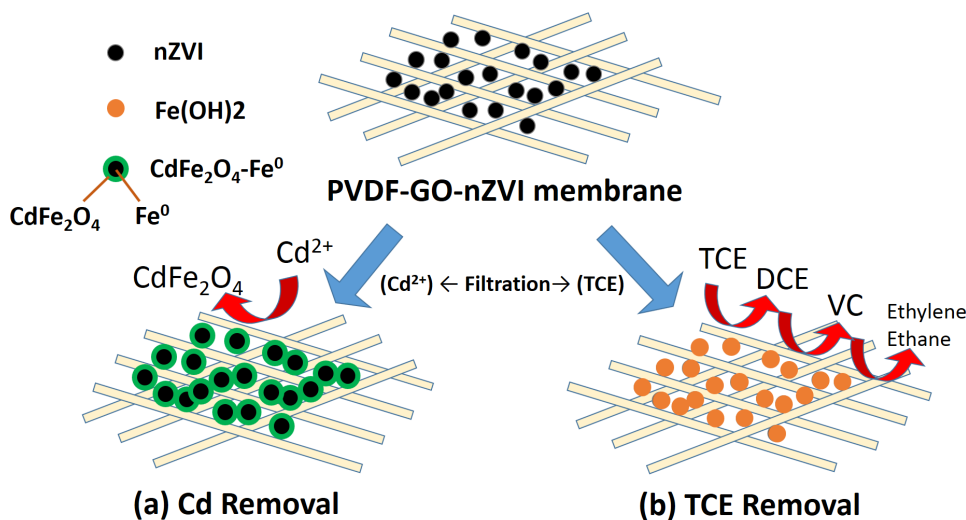
363 In terms of TCE filtration test (Fig. 13), the flux of each membrane (Fig. 13a) was
 364 obviously higher than the flux in Cd(II) filtration test (Fig. 12a). Meanwhile, the flux decline
 365 of each nZVI membrane in TCE filtration was slower than those in Cd filtration. These could
 366 be respectively explained by the followed two reasons: (1) TCE itself is a hydrophobic liquid

367 (Kiruthika et al., 2014), which has a similar hydrophobicity with the PVDF fibers (Woo et al.,
368 2016). Therefore, it had a higher flux for TCE than Cd(II) solution (2) In contrast to the
369 adsorption of nZVI to Cd(II), the mechanism of TCE removal was the dechlorination process
370 (**Fig. 14b**), which means TCE would not deposit on the surface of nZVI particles. It is reported
371 that TCE was first broken by electron attack originating from the surface of nZVI, and
372 dichloroethylenes (DCE) are formed as intermediates (Tian et al., 2018). Subsequently, DCE
373 was dechlorinated to vinyl chloride (VC) and eventually all dichloroethylene formed was
374 completely dechlorinated to ethylene and ethane (Hara et al., 2005). nZVI was reduced to
375 Fe(OH)₂ in the anaerobic groundwater ambience (Tian et al., 2018). Therefore, TCE molecule
376 would not adsorb on nZVI surface and hence remained higher pore sizes, fluxes and removal.
377 Particularly, PVDF-GO3-nZVI with the largest mean pore size had the highest initial flux at
378 317 LMH and PVDF-nZVI with the smallest mean pore size had the lowest initial flux at 123
379 LMH. It is noticed that the flux of PVDF-GO1-nZVI is just a bit lower than that of PVDF-
380 GO3-nZVI but much higher than those of PVDF-GO0.5-nZVI and PVDF-nZVI. Nevertheless,
381 the PVDF-GO3-nZVI had a higher pore size, its flux is still limited by the undesirable
382 aggregation of nZVI. On the other hand, the TCE removals of all the membranes (**Fig. 13b**)
383 were obviously lower than their removals for Cd(II) (**Fig.12b**). This may be caused by two
384 reasons: 1) the higher flux of membrane made TCE molecules have less hydraulic retention
385 time to contact and react with nZVI particles. 2) The dechlorination process of TCE was multi-
386 steps (**Fig.14b**), which had a lower total reaction rate than Cd(II) and requires more reaction
387 time. Specifically, PVDF-nZVI had the lowest removal no more than 4% over the whole
388 filtration test and PVDF-GO0.5-nZVI had a very little removal lower than 10% after 100ml.
389 In contrast, PVDF-GO1-nZVI had the highest removal at 82% at the 100 ml. Although PVDF-
390 GO3-nZVI had a slightly higher flux over 500 ml, PVDF-GO1-nZVI with the overwhelming

391 removal had a better performance in the filtration test. Generally, PVDF-GO1-nZVI had the
 392 best performance in gravity-driven filtration test for either Cd(II) removal or TCE removal.



393
 394 **Figure 13.** Performances of PVDF-GO-nZVI electrospun nanofiber membranes in gravity-
 395 driven membrane filtration test for TCE removal: (a) Flux and (b) TCE removal ratio
 396



397
 398 **Figure 14.** Mechanism of (a) Cd²⁺ and (b) TCE removal by PVDF-GO-nZVI electrospun
 399 nanofiber membranes
 400

401 **4. Concluding remarks**

402 In this study, we successfully fabricated PVDF-GO-nZVI membranes via
403 electrospinning technique and developed nZVI electrospun nanofiber membranes were tested
404 with two typical nZVI-targeted contaminants (Cd(II) and TCE) via gravity-driven membrane
405 filtration application. The main conclusions are listed as below:

- 406 ➤ GO can introduce functional groups (-COOH) to immobilize nZVI particles as well as
407 improve membrane hydrophilicity to increase filtration flux.
- 408 ➤ The more ratio of GO added in PVDF membrane, the more nZVI can be immobilized, but
409 too high GO ratio could make an uneven fiber size distribution in electrospinning process
410 and hence can lead to nZVI aggregation.
- 411 ➤ For Cd removal, as the adsorption mechanism made the Cd(II) deposited on the surface of
412 nZVI, PVDF-GO1-nZVI had a better nZVI distribution against the agglomeration, leading
413 to a better removal (100%) and flux (255 LMH).
- 414 ➤ For TCE removal, although PVDF-GO3-nZVI had a slightly higher flux (317 LMH),
415 PVDF-GO1-nZVI with the overwhelming removal (82%) had a better performance in the
416 filtration test.

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422 **References**

- 423 Bichara DA, Bodugoz-Sentruk H, Ling D, Malchau E, Bragdon CR, Muratoglu OK.
424 Osteochondral defect repair using a polyvinyl alcohol-polyacrylic acid (PVA-PAAc)
425 hydrogel. *Biomedical Materials* 2014; 9: 045012.
- 426 Boo C, Lee J, Elimelech M. Omniphobic polyvinylidene fluoride (PVDF) membrane for
427 desalination of shale gas produced water by membrane distillation. *Environmental*
428 *science & technology* 2016; 50: 12275-12282.
- 429 Boparai HK, Joseph M, O'Carroll DM. Cadmium (Cd 2+) removal by nano zerovalent iron:
430 surface analysis, effects of solution chemistry and surface complexation modeling.
431 *Environmental Science and Pollution Research* 2013; 20: 6210-6221.
- 432 Dong H, Guan X, Lo IM. Fate of As (V)-treated nano zero-valent iron: determination of arsenic
433 desorption potential under varying environmental conditions by phosphate extraction.
434 *Water research* 2012; 46: 4071-4080.
- 435 Dong H, Zhao F, He Q, Xie Y, Zeng Y, Zhang L, et al. Physicochemical transformation of
436 carboxymethyl cellulose-coated zero-valent iron nanoparticles (nZVI) in simulated
437 groundwater under anaerobic conditions. *Separation and Purification Technology* 2017;
438 175: 376-383.
- 439 Ghaffar A, Zhang L, Zhu X, Chen B. Porous PVdF/GO Nanofibrous Membranes for Selective
440 Separation and Recycling of Charged Organic Dyes from Water. *Environmental*
441 *science & technology* 2018; 52: 4265-4274.
- 442 Gu M, Farooq U, Lu S, Zhang X, Qiu Z, Sui Q. Degradation of trichloroethylene in aqueous
443 solution by rGO supported nZVI catalyst under several oxic environments. *Journal of*
444 *hazardous materials* 2018; 349: 35-44.

445 Hara J, Ito H, Suto K, Inoue C, Chida T. Kinetics of trichloroethene dechlorination with iron
446 powder. *Water Research* 2005; 39: 1165-1173.

447 Hidalgo KTS, Guzmán-Blas R, Ortiz-Quiles EO, Fachini ER, Corchado-García J, Larios E, et
448 al. Highly organized nanofiber formation from zero valent iron nanoparticles after
449 cadmium water remediation. *Rsc Advances* 2015; 5: 2777-2784.

450 Hu M, Zheng S, Mi B. Organic fouling of graphene oxide membranes and its implications for
451 membrane fouling control in engineered osmosis. *Environmental science & technology*
452 2016; 50: 685-693.

453 Hwang Y, Lee Y-C, Mines PD, Huh YS, Andersen HR. Nanoscale zero-valent iron (nZVI)
454 synthesis in a Mg-aminoclay solution exhibits increased stability and reactivity for
455 reductive decontamination. *Applied Catalysis B: Environmental* 2014; 147: 748-755.

456 Issa AA, Mariam Al Ali S, Mrlík M, Luyt AS. Electrospun PVDF graphene oxide composite
457 fibre mats with tunable physical properties. *Journal of Polymer Research* 2016; 23: 232.

458 Jang W, Yun J, Jeon K, Byun H. PVdF/graphene oxide hybrid membranes via electrospinning
459 for water treatment applications. *RSC Advances* 2015; 5: 46711-46717.

460 Kang G-d, Cao Y-m. Application and modification of poly (vinylidene fluoride)(PVDF)
461 membranes—a review. *Journal of Membrane Science* 2014; 463: 145-165.

462 Kim H, Hong H-J, Jung J, Kim S-H, Yang J-W. Degradation of trichloroethylene (TCE) by
463 nanoscale zero-valent iron (nZVI) immobilized in alginate bead. *Journal of hazardous*
464 *materials* 2010; 176: 1038-1043.

465 Kiruthika S, Gupta R, Rao K, Chakraborty S, Padmavathy N, Kulkarni GU. Large area solution
466 processed transparent conducting electrode based on highly interconnected Cu wire
467 network. *Journal of Materials Chemistry C* 2014; 2: 2089-2094.

468 Lefevre E, Bossa N, Wiesner MR, Gunsch CK. A review of the environmental implications of
469 in situ remediation by nanoscale zero valent iron (nZVI): behavior, transport and

470 impacts on microbial communities. *Science of the Total Environment* 2016; 565: 889-
471 901.

472 Liu C, Li X, Ma B, Qin A, He C. Removal of water contaminants by nanoscale zero-valent
473 iron immobilized in PAN-based oxidized membrane. *Applied Surface Science* 2014;
474 321: 158-165.

475 Ma H, Huang Y, Shen M, Guo R, Cao X, Shi X. Enhanced dechlorination of trichloroethylene
476 using electrospun polymer nanofibrous mats immobilized with iron/palladium
477 bimetallic nanoparticles. *Journal of hazardous materials* 2012; 211: 349-356.

478 Manavi - Tehrani I, Rabiee M, Parviz M, Tahriri MR, Fahimi Z. Preparation, Characterization
479 and Controlled Release Investigation of Biocompatible pH - Sensitive PVA/PAA
480 Hydrogels. *Macromolecular symposia*. 296. Wiley Online Library, 2010, pp. 457-465.

481 Ren J, Tijing LD, Shon HK. "Robbing behavior" and re-immobilization of nanoscale zero-
482 valent iron (nZVI) onto electrospun polymeric nanofiber mats for trichloroethylene
483 (TCE) remediation. *Separation and Purification Technology* 2017a; 189: 375-381.

484 Ren J, Woo YC, Yao M, Tijing LD, Shon HK. Enhancement of nanoscale zero-valent iron
485 immobilization onto electrospun polymeric nanofiber mats for groundwater
486 remediation. *Process Safety and Environmental Protection* 2017b; 112: 200-208.

487 Ren J, Yao M, Woo YC, Tijing LD, Kim J-H, Shon HK. Recyclable nanoscale zerovalent iron
488 (nZVI)-immobilized electrospun nanofiber composites with improved mechanical
489 strength for groundwater remediation. *Composites Part B: Engineering* 2019.

490 Sharma AK, Kumar R, Mittal S, Hussain S, Arora M, Sharma RC, et al. In situ reductive
491 regeneration of zerovalent iron nanoparticles immobilized on cellulose for atom
492 efficient Cr (VI) adsorption. *RSC Advances* 2015; 5: 89441-89446.

493 Şinoforoğlu M, Gür B, Arık M, Onganer Y, Meral K. Graphene oxide sheets as a template for
494 dye assembly: graphene oxide sheets induce H-aggregates of pyronin (Y) dye. Rsc
495 Advances 2013; 3: 11832-11838.

496 Thakur VK, Tan EJ, Lin M-F, Lee PS. Poly (vinylidene fluoride)-graft-poly (2-hydroxyethyl
497 methacrylate): a novel material for high energy density capacitors. Journal of Materials
498 Chemistry 2011; 21: 3751-3759.

499 Tian H, Liang Y, Zhu T, Zeng X, Sun Y. Surfactant-enhanced PEG-4000-NZVI for
500 remediating trichloroethylene-contaminated soil. Chemosphere 2018; 195: 585-593.

501 Tosco T, Papini MP, Viggi CC, Sethi R. Nanoscale zerovalent iron particles for groundwater
502 remediation: a review. Journal of Cleaner Production 2014; 77: 10-21.

503 Vidmar J, Oprčkal P, Milačić R, Mladenović A, Ščančar J. Investigation of the behaviour of
504 zero-valent iron nanoparticles and their interactions with Cd²⁺ in wastewater by single
505 particle ICP-MS. Science of The Total Environment 2018; 634: 1259-1268.

506 Wan H, Briot NJ, Saad A, Ormsbee L, Bhattacharyya D. Pore functionalized PVDF membranes
507 with in-situ synthesized metal nanoparticles: Material characterization, and toxic
508 organic degradation. Journal of membrane science 2017; 530: 147-157.

509 Wang C, Luo H, Zhang Z, Wu Y, Zhang J, Chen S. Removal of As (III) and As (V) from
510 aqueous solutions using nanoscale zero valent iron-reduced graphite oxide modified
511 composites. Journal of Hazardous materials 2014a; 268: 124-131.

512 Wang X, Chen C, Liu H, Ma J. Preparation and characterization of PAA/PVDF membrane-
513 immobilized Pd/Fe nanoparticles for dechlorination of trichloroacetic acid. Water
514 research 2008; 42: 4656-4664.

515 Wang Z, Cui Y, Wang J, Yang X, Wu Y, Wang K, et al. The effect of thick fibers and large
516 pores of electrospun poly (ε-caprolactone) vascular grafts on macrophage polarization
517 and arterial regeneration. Biomaterials 2014b; 35: 5700-5710.

518 Woo YC, Kim Y, Shim W-G, Tijing LD, Yao M, Nghiem LD, et al. Graphene/PVDF flat-sheet
519 membrane for the treatment of RO brine from coal seam gas produced water by air gap
520 membrane distillation. *Journal of Membrane Science* 2016; 513: 74-84.

521 Woo YC, Tijing LD, Park MJ, Yao M, Choi J-S, Lee S, et al. Electrospun dual-layer nonwoven
522 membrane for desalination by air gap membrane distillation. *Desalination* 2017; 403:
523 187-198.

524 Yao M, Ren J, Akther N, Woo YC, Tijing LD, Kim S-H, et al. Improving membrane distillation
525 performance: morphology optimization of hollow fiber membranes with selected non-
526 solvent in dope solution Revised manuscript for *Chemosphere*. *Chemosphere* 2019.

527 Zeng Z, Yu D, He Z, Liu J, Xiao F-X, Zhang Y, et al. Graphene oxide quantum dots covalently
528 functionalized PVDF membrane with significantly-enhanced bactericidal and
529 antibiofouling performances. *Scientific reports* 2016; 6: 20142.