

## Research Article

# Highly Sensitive H<sub>2</sub> Sensors Based on Co<sub>3</sub>O<sub>4</sub>/PEI-CNTs at Room Temperature

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The highly dispersed Co<sub>3</sub>O<sub>4</sub> on the surface of CNTs modified with polyethylenimine (PEI) was synthesized using the hydrothermal method. In the CNT-Co<sub>3</sub>O<sub>4</sub> composite materials, CNTs not only provide the substrate for the Co<sub>3</sub>O<sub>4</sub> nanoparticles but also prevent their aggregation. Furthermore, the interaction between Co<sub>3</sub>O<sub>4</sub> and CNTs modified with polyethylenimine (PEI) helps to improve the gas sensing performance. In particular, the CNT-Co<sub>3</sub>O<sub>4</sub> composite synthesized at 190°C shows the outstanding sensitive characteristics to H<sub>2</sub> with a lower detection limit of 30 ppm at room temperature. The obtained CNT-Co<sub>3</sub>O<sub>4</sub> sensor displays excellent selectivity and stability to H<sub>2</sub>. The energy band model of the conductive mechanism has been built to explain the resistance change when the gas sensor is exposed to the H<sub>2</sub>. Hence, the CNT-Co<sub>3</sub>O<sub>4</sub> composite material presents highly promising applications in H<sub>2</sub> gas sensing.

## 1. Introduction

Hydrogen energy is the most ideal clean and renewable energy with high heating value, so it is widely used in spacecraft, fuel cell, and internal combustion engine fuel [1–4]. Hydrogen can be stored in the form of gas, liquid, or even solid metal hydrides; thus, the hydrogen can be very potentially convenient to be employed. However, hydrogen gas is colorless, odorless, highly volatile, and flammable. Explosion is very easy to occur when hydrogen gas concentration exceeds 4% in dry air; therefore, safety has always been a big issue in hydrogen storage and use [5, 6]. Developing a sensor for hydrogen gas is essential to monitor its amount for their applications in energy and environmental fields.

Metal oxide semiconductors in nanostructure have been widely used as the active materials for gas sensing because of their high catalytic activities and improved selectivity from the distinctive structure [7–10]. Cobalt oxide (Co<sub>3</sub>O<sub>4</sub>) is a *p*-type semiconductor and could be potentially employed for supercapacitors, electrochemical devices, and gas sensors

due to its excellent electrocatalytic properties, high biocompatibility, and low cost [11–16]. A variety of harmful and toxic gases were detected using Co<sub>3</sub>O<sub>4</sub> nanoparticles, such as volatile organic compounds (VOCs), H<sub>2</sub>S, and NH<sub>3</sub> [17–20]. To improve the catalytic effectiveness and stability of Co<sub>3</sub>O<sub>4</sub> nanoparticles, the design of novel nanostructures and hybrid nanomaterials based on Co<sub>3</sub>O<sub>4</sub> nanoparticles is highly in demand.

Carbon nanotubes (CNTs) have shown the effects on the enhancement in catalytic activity for metal oxide gas sensing owing to the distinguished physical and chemical properties, such as excellent mechanical strength, thermal stability, hydrophilicity and stability, and electrical conductivity [21–24], because of their unique surface structures. There are many oxygen-containing groups (–COOH), graphite, and defects on the surface structures of CNTs from acidized treatment [25–27]. Zhang and coworkers showed that the CNTs could reduce the reaction temperature and enhance the gas sensitivity to CO gas for a metal oxide gas sensing system [28, 29]. Nguyen synthesized CNT films via CVD

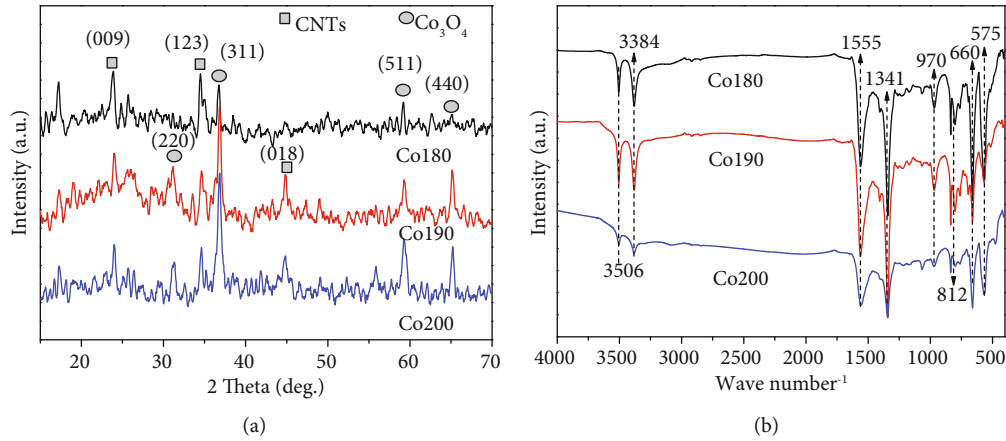


FIGURE 1: (a) XRD patterns of CNT- $\text{Co}_3\text{O}_4$  composites. (b) FT-IR spectra of CNT- $\text{Co}_3\text{O}_4$  composites synthesized by a hydrothermal method at different temperatures.

and displayed fast response and high sensitivity for  $\text{NH}_3$  sensing [30, 31]. CNTs also have been applied in the  $\text{NO}_2$  gas sensor, and the  $\text{NO}_2$  surface reaction can be explained by  $\text{NO}$  and  $\text{NO}_3$  which are produced on the surface [32–34].

In this work,  $\text{Co}_3\text{O}_4$  nanoparticles were successfully loaded onto the polyethylenimine (PEI) modified CNTs via a hydrothermal method [28, 35]. The functional groups on the sidewalls of CNTs benefited from the binding of  $\text{Co}^{2+}$  ions or  $\text{Co}_3\text{O}_4$  nanoparticles onto the CNTs. The  $\text{Co}_3\text{O}_4$  nanoparticles were highly dispersed with the PEI-CNTs displaying high conductivity and uniform contact with hydrogen. The prepared hybrid material was integrated on the surface of a planar interdigitated gold electrode for hydrogen detection, and the results showed that PEI-CNTs did improve the gas sensing performance. The composite gas sensor exhibits perfect sensitivity, selectively, and response speed upon exposure to  $\text{H}_2$  at room temperature.

## 2. Experiment

**2.1. Syntheses of CNT- $\text{Co}_3\text{O}_4$  Nanomaterials.** The CNT- $\text{Co}_3\text{O}_4$  composite materials were synthesized via a hydrothermal method. CNTs (Shenzhen Nanotech Port Co. Ltd., diameter: 30–80 nm, length: 5–30  $\mu\text{m}$ ) were functionalized firstly by acid treatment in the solution ( $\text{HNO}_3:\text{H}_2\text{SO}_4$  in a 1:3 (v/v) ratio). 80 mL of 0.03 mg/mL CNTs was added in 20 mL of 2 mg/mL PEI (branched PEI, Mw = 600, Cheng Du Micxy Chemical Co., Ltd.) under magnetic stirring, and the pH value of the solution was kept at 9.0. Subsequently, 20 mL of 2 mg/mL  $\text{Co}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$  (Tianjin Regent Chemical, Tianjin, China) was added to the above mixed solution during stirring. After that, saturated sodium hydroxide solution was slowly added into the mixed solution to adjust the pH to 13. The air flow was set to the solution with 50 mL/min for 24 h. Then, the mixed solution was exfoliated by ultrasonication at a power of 100 W for 1 h, and then, the mixed solution was placed still for 24 h at room temperature. A black precipitate was filtered and washed with distilled water to neutral. Lastly, the precipitate was dispersed in distilled water and poured into the Teflon-lined stainless steel autoclave. Hydrothermal synthesis was conducted at 180,

190, and 200°C for 3 h, respectively. The obtained CNT- $\text{Co}_3\text{O}_4$  composites were named as Co180, Co190, and Co200, respectively.

**2.2. Characterization.** The crystalline structures of the products was characterized by X-ray powder diffraction (XRD, D/max-III B-40 kV, Japan, Cu-K $\alpha$  radiation,  $\lambda = 1.5406 \text{ \AA}$ ). The Fourier transform infrared (FT-IR) spectra were acquired with the FT-IR spectrometer (PerkinElmer Spectrometer, KBr pellet technique). The morphology of the synthesized samples were investigated using TEM (JEOL-JEM-2100, 200 kV).

**2.3. Sensor Fabrication and Electrochemical Performance Measurements.** The CNT- $\text{Co}_3\text{O}_4$  materials were dispersed in ethanol to form a suspension. The suspension was drop-casted to cover the surface of a planar interdigitated gold electrode on a ceramic substrate. Then, the electrode was dried in a vacuum oven at 80°C for 3 h. The electrochemical performances of the sensor were measured in a chamber with a gas flow apparatus. The electrical resistance of the sensor was measured using a computer-controlled multimeter for filtering signals (50 Hz  $\pm$  5%, Shanghai, China). During the gas sensing test, a certain volume of  $\text{H}_2$  (99.9%, Dalian Great Special Gas Co., Ltd.) was injected with air at room temperature (25°C) in a relative humidity (RH) of around 45%. The response of the sensor was defined as the change of resistance in the two different gas circumstances:

$$S = \frac{R_g}{R_a}, \quad (1)$$

where  $R_g$  is the resistance in the tested gas and  $R_a$  is the resistance in air.

## 3. Results and Discussion

**3.1. Structure Characterizations.** The XRD diffraction and the FT-IR spectra of CNT- $\text{Co}_3\text{O}_4$  composites are shown in Figures 1(a) and 1(b), respectively. In the XRD patterns shown in Figure 1(a), two types of peaks can be indexed.

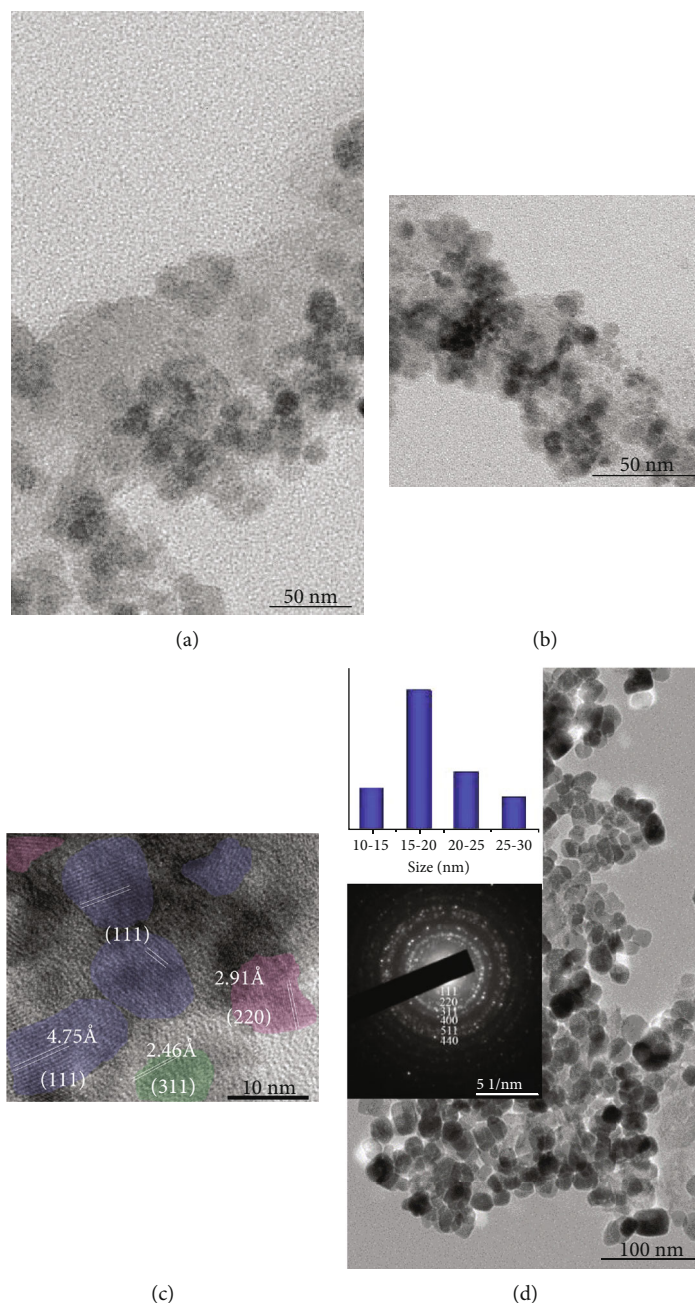


FIGURE 2: TEM characterization of CNT-s $\text{Co}_3\text{O}_4$  composite materials fabricated via the hydrothermal method at 180, 190, and 200°C: low-magnification and high-resolution TEM images of Co180 (a) and Co190 (b, c) and low-magnification TEM image and SAED of Co200 (d).

Diffraction peaks at 23.9°, 34.3°, and 44.7° can be indexed as the (009), (123), and (018) planes of the functional CNTs (JCPDS no. 74-2328, JCPDS no. 50-1086), and the d-spacings were calculated as 3.72, 2.61, and 2.03 Å, respectively. The XRD spectra showed a face-centered cubic  $\text{Co}_3\text{O}_4$  phase with the corresponding peaks at 31.2°, 36.8°, 59.3°, and 65.2° (JCPDS no. 42-1467) to the planes of (220), (311), (511), and (440). Furthermore, there are no obvious (220) and (440) peaks existing in the Co180. From the XRD of CNT- $\text{Co}_3\text{O}_4$ , we can find that the width at the half peak height of the same crystal plane is also different, indicating that the crystallinity and grain size of  $\text{Co}_3\text{O}_4$  were influenced by the hydrothermal temperature. In addition,

the high purity of the CNT- $\text{Co}_3\text{O}_4$  composites can be proven because of the two types of peaks.

The FT-IR spectra of CNT- $\text{Co}_3\text{O}_4$  composite materials are shown in Figure 1(b). The broad peak around 3506 and 3384  $\text{cm}^{-1}$  would be attributed to the N-H stretch of secondary and primary amines of PEI in samples [36]. The peak at 1557  $\text{cm}^{-1}$  is attributed to the C-C stretching vibration in CNTs [37], suggesting the CNTs presenting in the nanocomposites.  $\text{CH}_2$  wagging from 1100 to 1500  $\text{cm}^{-1}$  is a typical mode of PEI [28]. The peak stretching from 990 to 910  $\text{cm}^{-1}$  and wagging from 840 to 800  $\text{cm}^{-1}$  can be attributed to the unsaturated hydrocarbon (=C-H) of CNTs. The sharp peaks at 575 and 660  $\text{cm}^{-1}$  are attributed to the

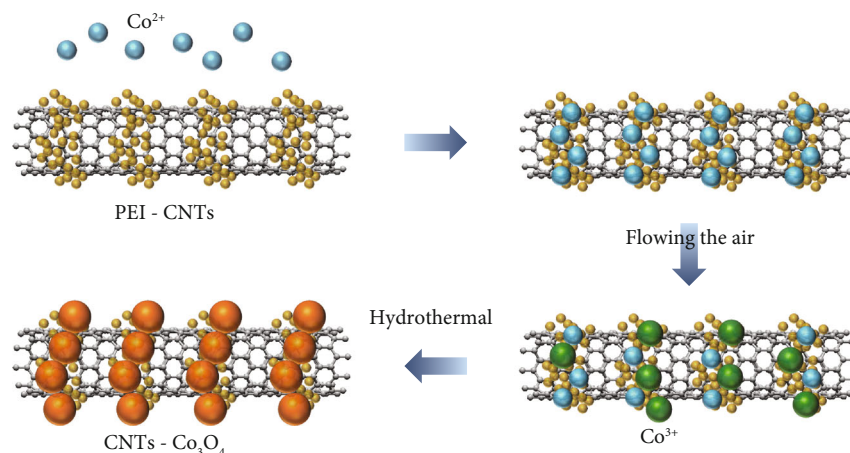


FIGURE 3: Scheme of the growth mechanism of the CNT- $\text{Co}_3\text{O}_4$  composite.

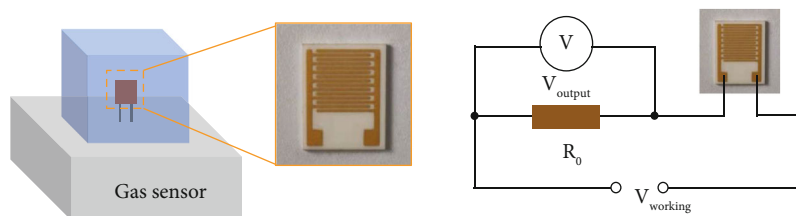


FIGURE 4: Sketches of the hydrogen sensor and the experimental circuit.

Co(III)-O and Co(II)-O stretching vibrations of  $\text{Co}_3\text{O}_4$  [38]. Hence, through the FT-IR characterization, it presents that PEI-CNTs and  $\text{Co}_3\text{O}_4$  nanoparticles have been successfully combined.

Figure 2 shows the TEM and HRTEM images of CNT- $\text{Co}_3\text{O}_4$  composite materials fabricated via the hydrothermal method at  $180^\circ\text{C}$ ,  $190^\circ\text{C}$ , and  $200^\circ\text{C}$ . Figures 2(a)–2(d) show that  $\text{Co}_3\text{O}_4$  nanoparticles were integrated onto the surface of CNTs, forming the CNT- $\text{Co}_3\text{O}_4$  composite materials with different densities of  $\text{Co}_3\text{O}_4$  nanoparticles on CNTs. Figure 2(a) displays relative loose distribution of the aggregated  $\text{Co}_3\text{O}_4$  nanoparticles on CNTs of the Co180 nanocomposite. Figure 2(b) shows that the aggregated  $\text{Co}_3\text{O}_4$  nanoparticles were densely distributed on the CNTs. The HRTEM image in Figure 2(c) further shows the dense  $\text{Co}_3\text{O}_4$  nanoparticles in different predominated crystal facets of (111), (220), and (311) with corresponding lattice spacing distance of 4.75, 2.91, and 2.46, respectively. The  $\text{Co}_3\text{O}_4$  nanoparticles obtained at  $190^\circ\text{C}$  presented relatively large size (10–12 nm), while at the hydrothermal temperature of  $200^\circ\text{C}$ , it can be found from Figure 2(d) that  $\text{Co}_3\text{O}_4$  nanoparticles almost fully covered the surface of CNTs. The SAED pattern of the Co200 nanocomposite displays the polycrystalline of  $\text{Co}_3\text{O}_4$  with crystal planes of (111), (220), (311), (400), (511), and (440). The size distribution of  $\text{Co}_3\text{O}_4$  nanoparticles in the Co200 nanocomposite was calculated, as shown in the inset of Figure 2(c). Relative large size distribution was found, and most of the  $\text{Co}_3\text{O}_4$  nanoparticles were within the range of 15–20 nm. We found that the hydrothermal synthesis temperature would be able to control the size of  $\text{Co}_3\text{O}_4$  nanoparticles and the coverage density

of  $\text{Co}_3\text{O}_4$  nanoparticles on CNT surfaces. Of the three synthetic temperatures, it shows the uniform size and morphology for the  $\text{Co}_3\text{O}_4$  nanoparticles formed at  $190^\circ\text{C}$ .

The hypothesis of the formation mechanism for the nanocomposites is proposed in Figure 3. The CNTs after acid treatment can be endowed a large amount of moieties of  $-\text{COOH}$  and  $-\text{OH}$ , which improves the dispersion of CNTs and enables the modification with PEI. In our work, the PEI would bond to the  $-\text{COO}^-$  groups of CNTs after both are mixed in the aqueous solution under pH 9; thus, PEI would be uniformly modified on the surface of the CNTs (step 1). The  $\text{Co}^{2+}$  from the precursor is easily dissociated and positively charged in aqueous solution. Then, we monitor the pH of the solution to 13 to unprotonate the PEI, so the  $\text{Co}^{2+}$  would be bonded to unprotonated PEI on the surface of CNTs and the  $\text{Co}^{2+}$  is functionalized onto CNT surfaces uniformly via PEI-CNTs (step 2). When air is introduced into the solution,  $\text{Co}^{2+}$  is oxidized to  $\text{Co}^{3+}$  gradually (step 3). The subsequent hydrothermal treatment at high temperature allows the in situ formation of  $\text{Co}_3\text{O}_4$  nanoparticles on the surface of CNTs (step 4).

**3.2. Gas Sensing Testing.** The response of CNT- $\text{Co}_3\text{O}_4$  materials to  $\text{H}_2$  gas was investigated through measuring the resistance versus time over two multifinger electrodes (size:  $10 \times 8 \times 0.15$  mm) at room temperature, as shown in Figure 4.

Figure 5 presents the  $\text{H}_2$  gas sensing performance of CNT- $\text{Co}_3\text{O}_4$  nanocomposite materials, Co180, Co190, and Co200, respectively. The concentration range of  $\text{H}_2$  gas was adjusted from 1000 ppm to 30 ppm. The nanocomposite of CNTs- $\text{Co}_3\text{O}_4$  can play a role as the  $p$ -type semiconductor with the hole as the carrier. When a  $p$ -type semiconductor contacts

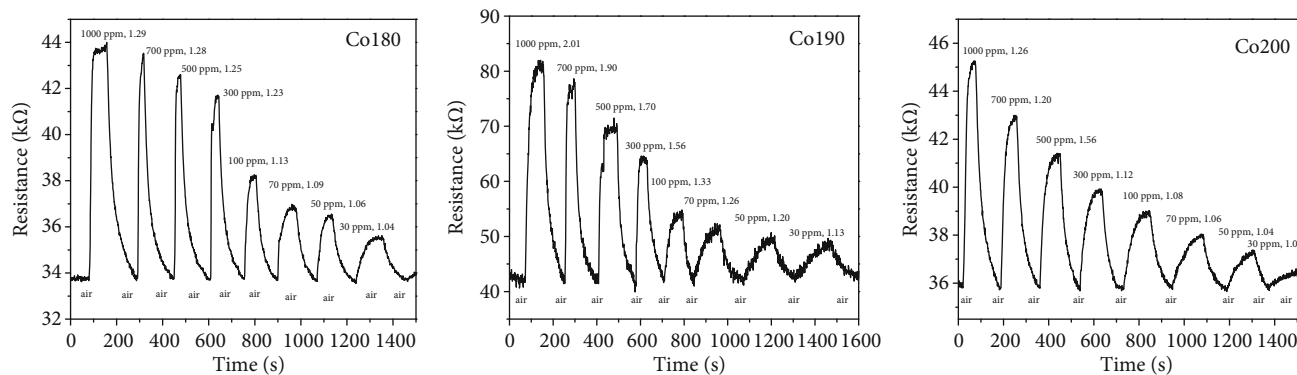


FIGURE 5: Dynamic response-recovery curves of the Co180, Co190, and Co200 thin film sensors to  $H_2$  at room temperature, respectively.

with the reducing gas, the reduction interaction between them would produce electrons and the  $p$ -type semiconductor would accept the electron; thus, the resistance of the semiconductor would be increased. In Figure 5, we can see that the resistance of the CNT- $Co_3O_4$  nanocomposite increased when the air with  $H_2$  passed through while its resistance dropped to the original value when air was introduced without  $H_2$ . The sensitivity decreases with the decrease in the concentration of the target gas, respectively, at room temperature. It is shown that the sensing response varies with the change in the concentration of the analyte gas.

By comparing the results shown in Figure 5, it can be seen that the response to  $H_2$  is significantly higher for Co190 than the response to  $H_2$  for Co180 and Co200 at room temperature. The response to the 1000 ppm  $H_2$  for the nanocomposite Co190 presents the highest resistance, 81.9 k $\Omega$ , which is 2.01 times higher than the original resistance value. The response of the Co190 to the  $H_2$  concentration of 30 ppm decreased, and the increased resistance is 1.13 times higher than the original value at room temperature. For the nanocomposite of Co200, the factor of the increased resistance decreased to 1.26 with 1000 ppm  $H_2$ . Hence, the nanocomposite of Co190 exhibited the highest sensing sensitivity among all three nanocomposite materials.

Furthermore, the CNT- $Co_3O_4$  nanocomposites fabricated at different hydrothermal temperatures displayed remarkably different effects on the recovery process of gas sensors. The possible reasons can be ascribed to the following aspects: (i) the  $Co_3O_4$  nanoparticles were better dispersed on the surface of CNTs at low temperature and the  $Co_3O_4$  nanoparticles were in relatively uniform size (shown in Figures 2(a) and 2(b)) and (ii) higher synthesis temperature caused large size of the  $Co_3O_4$  nanoparticles and wide size distribution. Moreover, higher hydrothermal temperature induced severe nanoparticles' aggregation and complete coverage on the surface of CNTs (shown in Figure 2(c)). This resulted in a low surface-area-to-volume ratio, which decreased the surface area of exposure to target gas for response.

The roles of the CNTs and PEI for the sensing performance of the nanocomposite materials were investigated as well. The sensing response to  $H_2$  of the pure  $Co_3O_4$  nanoparticles, nanocomposite of  $Co_3O_4$ -PEI without CNTs, and nanocomposite of  $Co_3O_4$ -CNTs without PEI is shown in

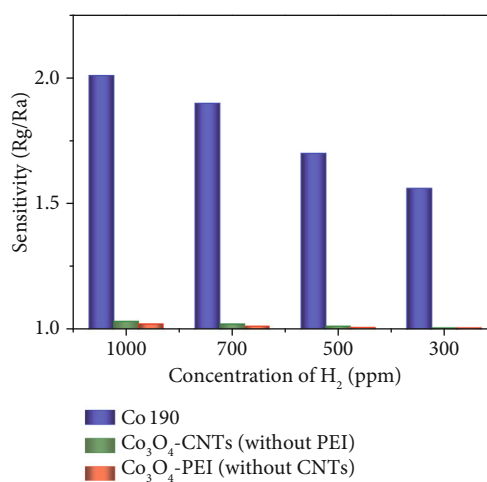


FIGURE 6: Comparison of sensitivities of the sensors using Co190,  $Co_3O_4$ -PEI (without CNTs), and  $Co_3O_4$ -CNTs (without PEI) to 300–1000 ppm  $H_2$  at room temperature.

Figure 6. It has been found that without CNTs or PEI in the nanomaterial, the sensor exhibited significantly lower response than that of the nanocomposite of  $Co_3O_4$ -CNTs with PEI. It could be related to the coupling effect between the Co and CNTs, which had been proposed to contribute to the enhanced oxygen reduction ability [28]. In the  $Co_3O_4$ /PEI-CNT composite, when the CNT architecture was employed as conducting scaffolds in a  $Co_3O_4$  semiconductor-based sensor, it not only prevents  $Co_3O_4$  nanoparticles from aggregation but can also boost the electron transfer efficiency. PEI provided high-density homogeneous functional groups on the CNTs' sidewalls for binding  $Co_3O_4$  nanoparticles. Meanwhile, PEI is helpful for high-density dispersion of  $Co_3O_4$  grains and enhances the interaction between  $Co_3O_4$  grains and CNTs and improves the transport of the carriers to the surface [39, 40]. Therefore, it could be concluded that the PEI played an important role in the sensing performance of the nanocomposites.

We also evaluated the sensing selectivity of the nanocomposites of CNT- $Co_3O_4$  to  $H_2$  with the interferences such as ethanol, methanol, hydrogen, benzene, and acetone, as shown in Figure 7(a). The response of the sensor to 1000 ppm  $H_2$  was

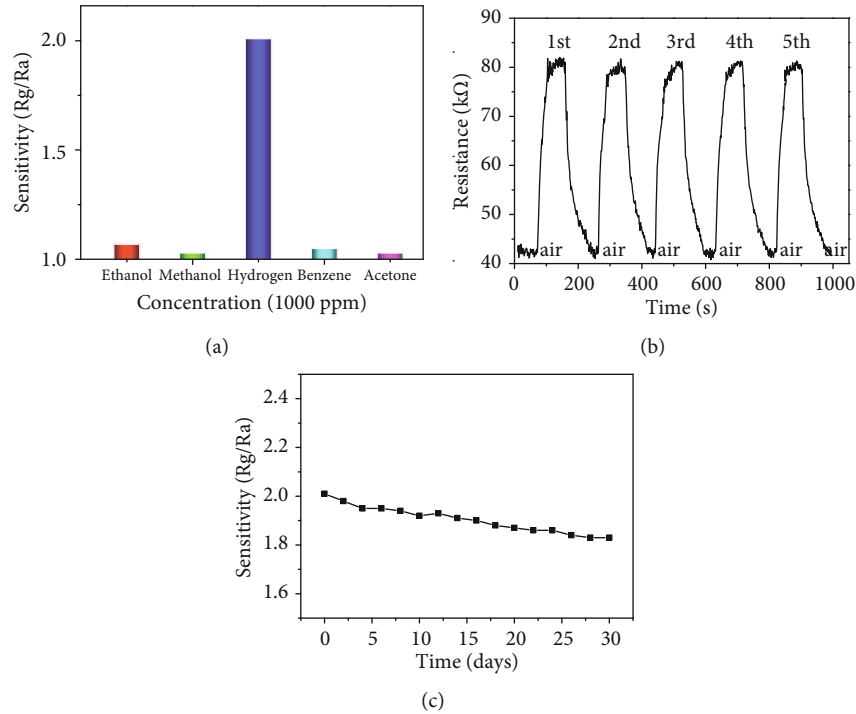


FIGURE 7: (a) Selectivity of the response of the CNT-Co<sub>3</sub>O<sub>4</sub> sensor to 1000 ppm of ethanol, methanol, hydrogen, benzene, and acetone at room temperature. (b) Cycling stability curves of dynamic response-recovery and (c) long-term stability response of the CNT-Co<sub>3</sub>O<sub>4</sub> sensor to 1000 ppm H<sub>2</sub> at room temperature.

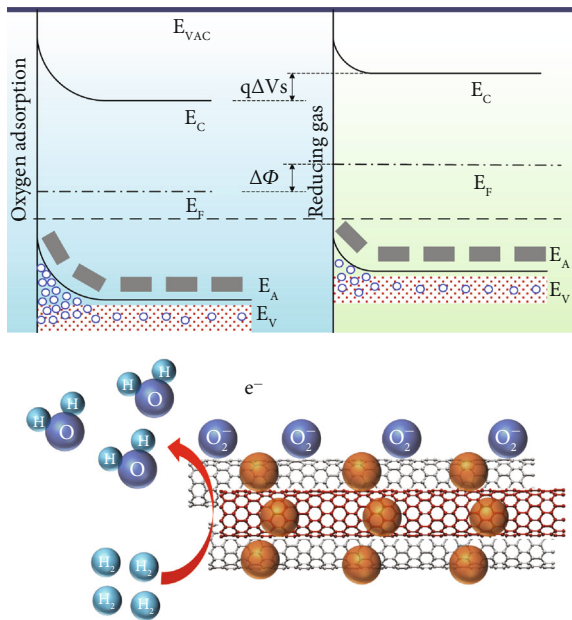


FIGURE 8: The structure and energy band model of the conductive mechanism in the air and reducing gas.

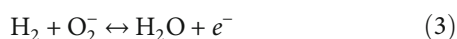
highest and 3–4 times higher than the responses to the other gases at the same concentration, suggesting an excellent selectivity to H<sub>2</sub> at room temperature.

To validate the sensing reliability and long-term stability of the CNT-Co<sub>3</sub>O<sub>4</sub> composite sensor, we repeated the measurement of the dynamic response-recovery five times and tested the sensibility for 30 days. The repeatability of

response and recovery curves shows the excellent repeatability in five consecutive experiments, as shown in Figure 7(b). After 30 days of long-term stability testing, the sensitivity of the sensor dropped from an initial 2.01 to 1.83; it retained 91.04% of its initial response, as shown in Figure 7(c). Therefore, CNTs-Co<sub>3</sub>O<sub>4</sub> displays potentials in a sensor and wide range of application prospects for an H<sub>2</sub> sensor.

**3.3. Gas Sensing Mechanism.** The gas sensing mechanism of the metal oxide nanoparticles is strongly related to surface reactions [41], and chemisorbed oxygen also makes important contribution to the sensing mechanism. In general, the chemisorbed oxygen species of O<sub>2</sub><sup>-</sup>, O<sup>-</sup>, and O<sup>2-</sup> can be formed at <150°C, 150~400°C, and >400°C, respectively. Because of the electrostatic interaction between the oppositely charged species, the adsorption of oxygen anions onto *p*-type oxide semiconductors induces the formation of hole accumulation layers. An energy band conduction model was used for the *p*-type oxide semiconductor as the gas sensors [7]. Figure 8 shows the structure and energy band model of the conductive mechanism in the air and reducing gas. When the *p*-type oxide semiconductor was exposed to the air, ambient oxygen can extract electrons from the valence band and form oxygen ions. It can increase the concentration of the holes, forming the hole accumulation layers. It can be depicted that the energy bands exhibited an upward band bending, of which length is the thickness of the hole accumulation layers. When the *p*-type oxide semiconductor is exposed to the reducing gas, the oxygen ions on the surface would be reacted. The decrease in the concentration of the holes, described in the energy band

representation as a downward band bending ( $q\Delta V_s = \Delta\Phi$  in Figure 8), results in the increase in the resistance of the gas sensor. Hence, in our study, the ambient oxygen is absorbed on the surface of the CNTs- $\text{Co}_3\text{O}_4$  and traps the electrons at the surface as the oxygen ions ( $\text{O}_2^-$ ) in the room temperature (2). When the CNT- $\text{Co}_3\text{O}_4$  was exposed under the ambient  $\text{H}_2$ , the reaction between reducing gas and oxygen ions ( $\text{O}_2^-$ ) can result in the increase in the resistance of the CNT- $\text{Co}_3\text{O}_4$  gas sensor (3).



#### 4. Conclusion

In summary, the CNT- $\text{Co}_3\text{O}_4$  nanocomposites with highly dispersed  $\text{Co}_3\text{O}_4$  were fabricated via the hydrothermal method. The CNT- $\text{Co}_3\text{O}_4$  nanocomposites display excellent sensing sensitivity, selectivity, and stability at room temperature to  $\text{H}_2$  gas. CNTs play multiple roles in the sensing performance, not only offering the  $\text{Co}_3\text{O}_4$  substrates to avoid its aggregation but also providing perfect migration of the electron pathway via the synergetic chemical coupling effect between  $\text{Co}_3\text{O}_4$  and CNTs. PEI also contributes to improving the performance of  $\text{H}_2$  sensors. Hence, the CNT- $\text{Co}_3\text{O}_4$  composites represent a potential application prospect in  $\text{H}_2$  detection.

#### Data Availability

The experimental data used to support the findings of this study are included within the article.

#### Conflicts of Interest

The authors declare that they have no conflicts of interest.

#### Acknowledgments

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