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Localized deposition of pure platinum nanostructures

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Abstract- Localized deposition of pure platinum nanostructures was achieved using a combination of focused electron beam induced processing (FEBID) of an inorganic platinum precursor and low temperature annealing in water vapour. This technique enables fabrication of Pt nanostructures with high spatial resolution and purity, for applications in nanoelectronics, sensing devices and catalysis.

Index Terms—Platinum, Nanostructured Materials, Scanning Electron Microscopy, Fabrication

I. INTRODUCTION

Focused electron beam induced deposition (FEBID) is a nanofabrication technique in which electrons are used to dissociate surface-adsorbed precursor molecules into volatile (by-product) and non-volatile (target deposition) components, typically under a constant partial pressure of a precursor gas [1]. This technique enables direct writing of high resolution (sub-nm) three-dimensional nanostructures without requiring the use of lithographic masks and resists.

The main limitation of FEBID has been the typically low purity of fabricated structures, which normally contain ≈ 50 at. % C and are composed of nano-sized metal grains in an amorphous carbon matrix [2]. Due to the high carbon content, EBID deposits behave as granular materials with percolative transport properties. A variety of methods have been used to improve the purity of FEBID deposits, including simultaneous and post-deposition annealing in a variety of gaseous atmospheres [1].

FEBID of Pt has attracted more attention than any other material. Platinum nanostructures are of particular interest for applications such as nanoelectronics (as nanowires for interconnects and contacts [3], nanodots for non-volatile memory devices), plasmonic antennas [4] and catalysis (in biosensors and fuel cells). In all of these applications, the resolution, surface area and purity of the deposited material are critical. FEBID of platinum has so far been conducted in high vacuum (HV) SEM using the standard metalorganic precursor methylcyclopentadienyl platinum (MeCpPtMe₃) in combination with post-deposition annealing in oxygen or a similar reactive gas [5]. However, thermal or electrical annealing of platinum nanowires fabricated by FEBID using metal organic precursors has been found to result in migration of Pt out of the nanowires, forming a Ptdepleted graphitic carbon deposit.

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In this work, we show that extremely high purity platinum can be deposited by FEBID using a carbon-free platinum precursor, Tetrakis(trifluorophosphine)platinum (Pt(PF₃)₄).

II. EXPERIMENTAL RESULTS

A. Platinum deposition process

The precursor used for Pt deposition is Tetrakis(triuorophosphine)platinum, $Pt(PF_3)_4$ (Strem chemicals), a high vapour pressure liquid at room temperature. Due to the sensitivity of this precursor to air and moisture, it was loaded into a peltier-cooled stainless steel bottle in a dry N_2 environment, and sealed using a manual valve. Multiple freeze-pump-thaw cycles were used to remove excess nitrogen and PF_3 gas, leaving behind only the platinum precursor.

During FEBID experiments, the precursor gas was flowed into a modified XL30 ESEM with an in-situ environmental reaction cell, previously described in [6]. The reaction cell is isolated from the main SEM chamber using a 200 micron pressure limiting aperture between the cell and the electron column. The differentially pumped reaction cell allows for both greater pressure control and a much larger ratio of precursor gas to residual molecules than is achievable using standard FEBID in HV SEMs.

During FEBID experimentation, the pressure in the reaction cell was kept constant at 0.1 Torr (13 Pa) using a pressure feedback system; the electron beam current was 10 nA; and the accelerating voltage was 10 kV. A defocused electron beam (beam diameter 0.6 mm) with a top-hat beam profile was used to dissociate the precursor, forming deposits with diameters of 5 μ m. A set of three 'defocused' deposits was fabricated by FEBID, with deposition times of 30 minutes each (Fig. 1a). Nanodot arrays can also be fabricated using very short beam dwell times (Fig. 1b). The defocused deposits are sufficiently large that the electron beam interaction volume is entirely contained within them, permitting compositional analysis by wavelength or energy dispersive x-ray spectroscopy (WDX or EDX).

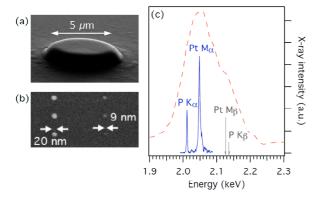


Fig 1. (a) Example of a typical defocused Pt deposit created specifically for compositional analysis. (b) High resolution Pt nano-dots. (c) WDX and EDX spectra of the same deposit region, highlighting the spectral resolution of each technique with regard to the Pt and P x-ray lines.

B. Compositional analysis

The composition of as-grown and annealed deposits was analyzed by WDX) (Fig. 2). This technique was chosen due to its' very high spectral resolution (< 10eV), which enables the normally overlapping P K and Pt M X-ray lines to be resolved. Deposits grown at room temperature have a typical atomic composition of $47 \pm 5\%$ Pt, $30 \pm 5\%$ P and $15 \pm 10\%$ O.

C. Post-deposition annealing

The trace amounts of fluorine and the presence of oxygen in the as-grown deposits indicate efficient dissociation and desorption of PF₃ and F, accompanied by oxidation of P by residual water present in the vacuum chamber. In order to further purify the material, post-deposition thermal annealing was performed in an ex-situ CVD chamber in a water vapour environment at a pressure of 1 torr (130 Pa). Deposits were annealed for 40-60 minutes in vacuum, and in H₂O, O₂ and NH₃ environments at a pressure of 130 Pa.

Annealing in vacuum, O_2 and NH_3 environments had no significant effect on composition. However, low temperature (400-600 °C) annealing in water vapour can increase the Pt content to the point where it is indistinguishable from reference, high purity Pt films by WDS analysis, as shown in Fig. 2. Temperatures of only 400 °C are required to obtain Pt purity of 94% after 300 min, with the remaining 5% being due to O and C impurities from adsorbed hydrocarbons and formation of a surface oxide layer during the WDS analysis.

III. CONCLUSION

We have reported a new method of fabricating highly pure, nanoscale Pt by combining FEBID with low temperature (<400 °C) post-deposition annealing in water vapour. The low annealing temperature results in little or no nanostructure deformation, which

can be a major problem with other FEBID methods for obtaining pure Pt and other materials.

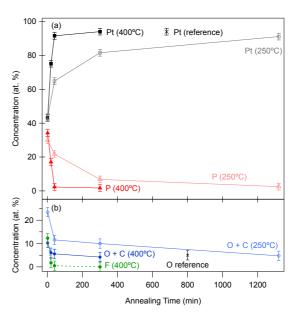


Fig 2. Composition of Pt(PF₃)₄-EBID deposits plotted as a function of annealing time in H₂O vapor at 250 and 400 °C. Fully purified Pt deposits are obtained after annealing for 300 min at 400 °C. The measured composition of a reference, high purity Pt film is also shown. [Adapted from Ref. 7]

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